

Anharmonic phonons in thermoelectrics and ferroelectrics studied with inelastic neutron and x-ray scattering

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<https://delaire.pratt.duke.edu/>

X-ray Echo Spectroscopy Workshop

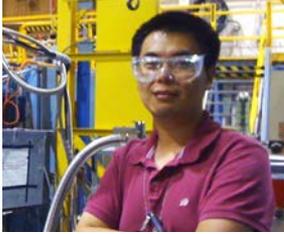
Advanced Photon Source

Sep 8-10, 2016

Acknowledgements

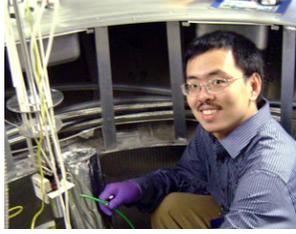
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ORNL: A. May, M. McGuire B. Sales (Sample Synthesis),
J. Budai, E. Specht (Scattering and ThermoPhysics Group)
Jiao Lin, V. Lynch, D. Singh, X. Chen, CAMM-group (Computing)
D. Abernathy, G. Ehlers, M. Stone, T. Hong, S. Chi, H. Cao (Neutron scattering)

ANL/APS: A. Said, A. Alatas, M. Hu, W. Bi

MIT: Gang Chen, Yang Shao-Horn

Linköping U.: Olle Hellman

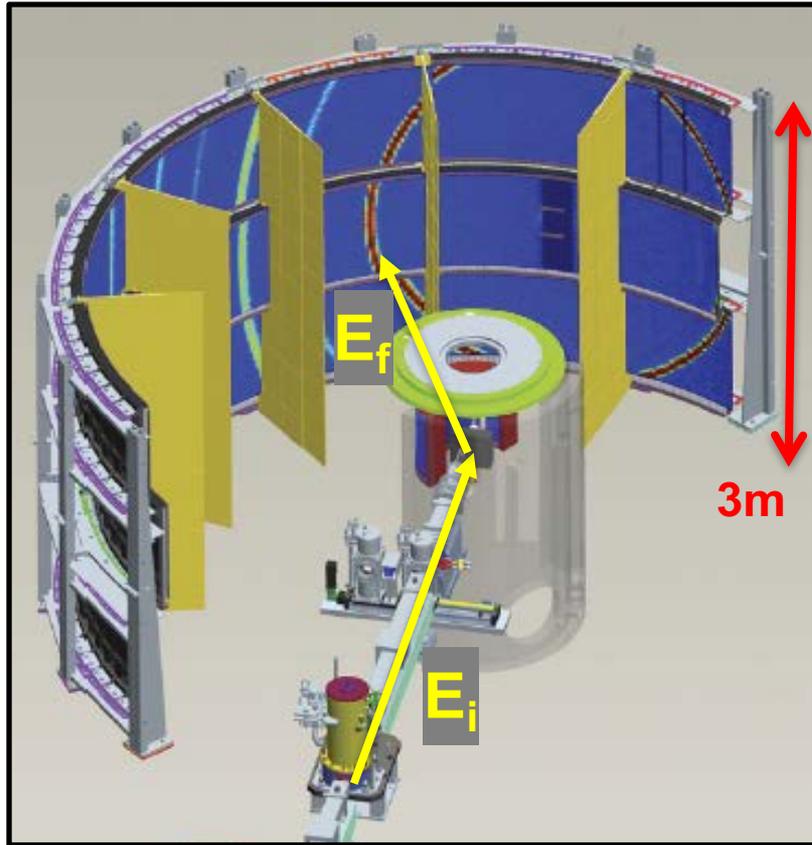
Funding:

- DOE Office of Science Early Career Award, DOE BES-MSED
- DOE Energy Frontier Research Center (S³TEC)
- Center for Accelerating Materials Modeling from SNS data (CAMM)

Neutron and X-ray Spectrometers

Neutron time-of-flight:

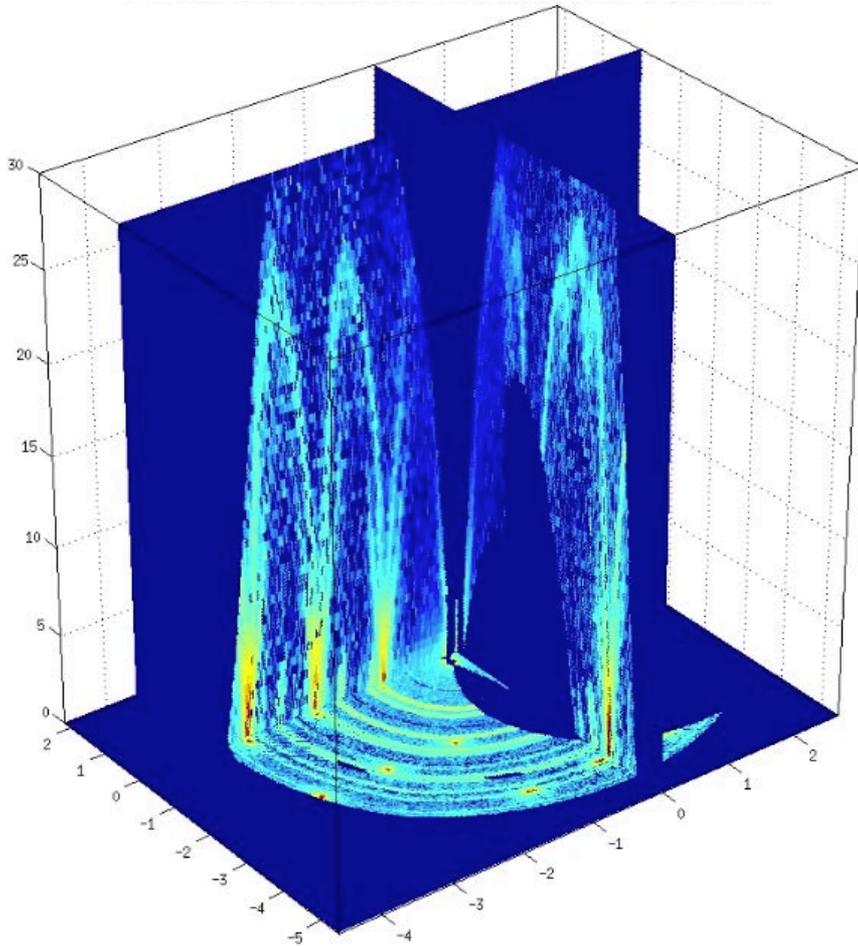
Inelastic X-ray Spectrometer:



ARCS at SNS: $\sim 10^5$ pixels,
 $\sim 10^4$ time channels each.
beam ~ 5 cm, $\Delta E/E \sim 5\%$

HERIX at APS: ~ 10 detectors.
beam $\sim 30 \mu\text{m}$; $E_i = 23.7 \text{ keV}$
Resolution $\sim 1.5 \text{ meV}$, constant with E

S(Q,E) mapping with INS



Niobium single crystal
ARCS, $E_i = 50$ meV, mass~10g
 $T = 300$ K

- With recent advances in time-of-flight chopper spectrometers, we can efficiently collect the entire 4-D (Q_x, Q_y, Q_z, E) space
- ARCS, CNCS, HYSPEC, SEQUOIA at SNS can provide these 4-D $S(Q, E)$ maps

***But requires large
single crystals!
> several grams***

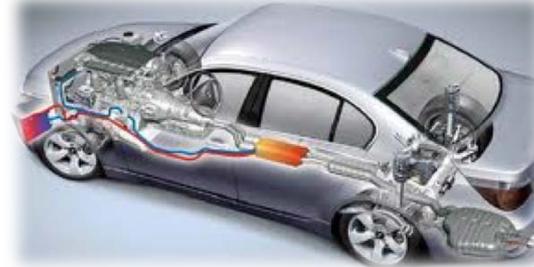
Phonon Transport in Thermoelectrics

- O. Delaire, J. Ma *et al.*, *Nature Materials* 10, 614 (2011)
- J. Ma*, O. Delaire*, *et al.*, *Nature Nanotechnology* 8, 445 (2013)
- C. Li, O. Hellman, O. Delaire *et al.*, *Phys. Rev. Letters* 112, 175501 (2014)
- C. Li, J. Ma, O. Delaire *et al.* *Phys. Rev. B* 90, 214303 (2014)
- Li*, Hong*, May, Bansal, Ma, Hong, Chi, Ehlers, and Delaire, *Nature Physics* 11, 1063 (2015)
- Bansal, Hong, Li, May, and Delaire *Phys. Rev. B* 94, 054307 (2016)

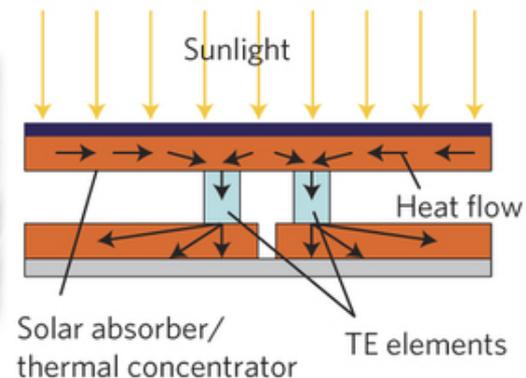
Radioisotope Thermoelectric Generators



Automobile industry



Solar thermoelectrics



Thermal conductivity in thermoelectrics

- Suppress lattice thermal conductivity κ_{lat} to optimize thermoelectric efficiency:

$$zT = \frac{(\sigma S^2 T)}{(\cancel{\kappa_{el}} + \kappa_{lat})}$$

S: Seebeck ; σ : el. conductivity

κ : thermal conductivity

- Rationalize *microscopic origins* of κ_{lat} to gain control over phonon transport:

$$\kappa_{q,j} = \frac{1}{3} C_{q,j} v_{q,j}^2 \tau_{q,j}$$

$\kappa = \sum_{q,j} \kappa_{q,j}$

Specific heat

Phonon group velocities

Phonon lifetimes: many possible scattering processes combined

All quantities can be obtained from neutron/x-ray scattering measurements and also computed from first-principles (for moderately complex cells).

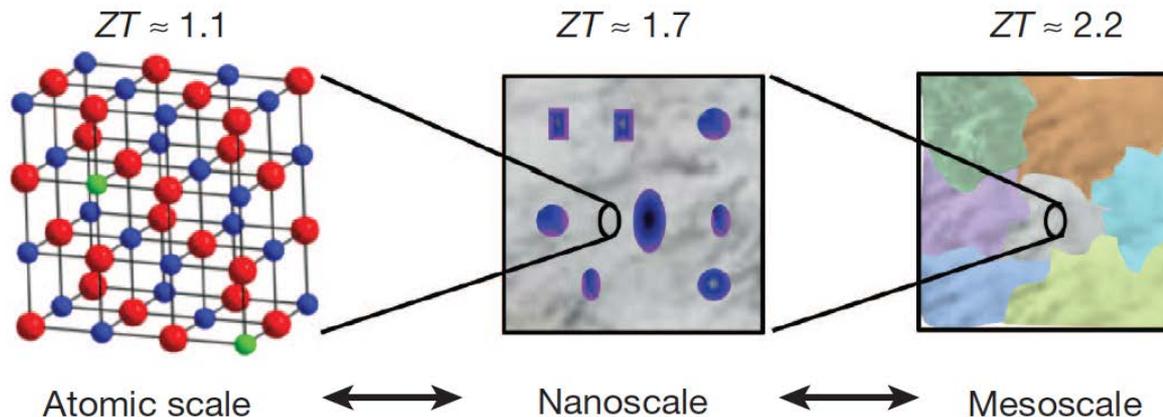
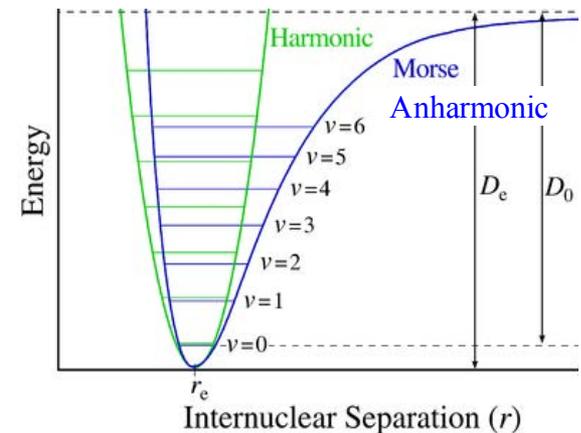
Phonons and Thermal Transport

- Perfect harmonic crystal would have *infinite thermal conductivity...*
- Phonons in real materials have finite lifetimes and mean-free-paths
- Phonon-phonon scattering from anharmonicity
- Other phonon scattering mechanisms: electron-phonon, spin-phonon, defects, nanostructure, mesoscale.

decay

mean free path

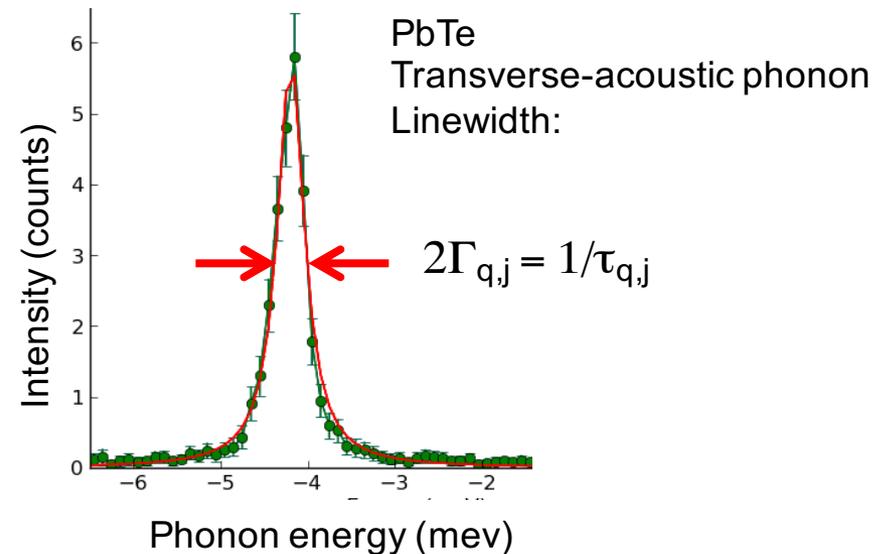
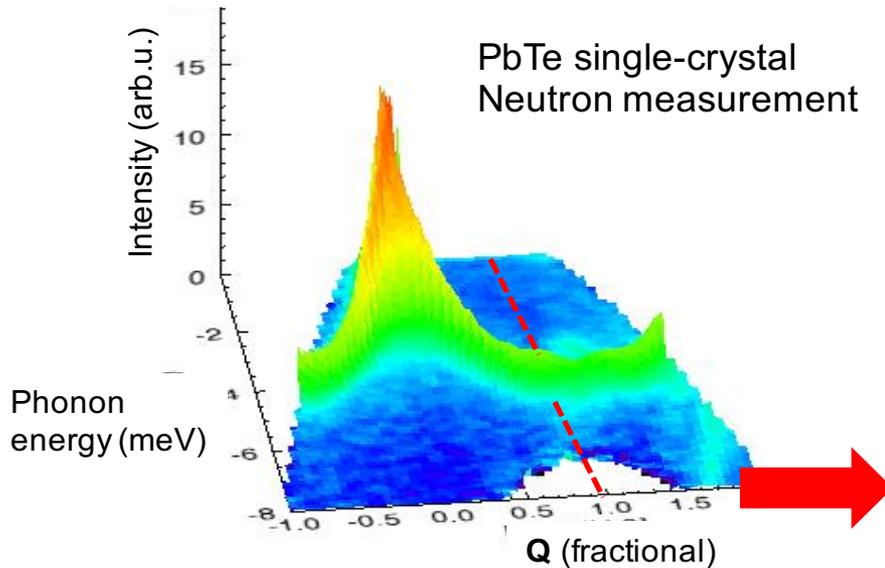
$$\Lambda_{q,j} = v_{q,j} \tau_{q,j}$$



Biswas *et al.* Nature 489, 414 (2012)

Thermal conductivity and INS measurements of phonon linewidths

INS / IXS measures **dynamical structure factor**, $S(\mathbf{Q}, E)$ providing details of dispersions and linewidths:



Estimate thermal conductivity from INS

$$\kappa = \sum_{q,j} \kappa_{q,j} = \frac{1}{3} \sum_{q,j} C_{q,j} v_{q,j}^2 \tau_{q,j}$$

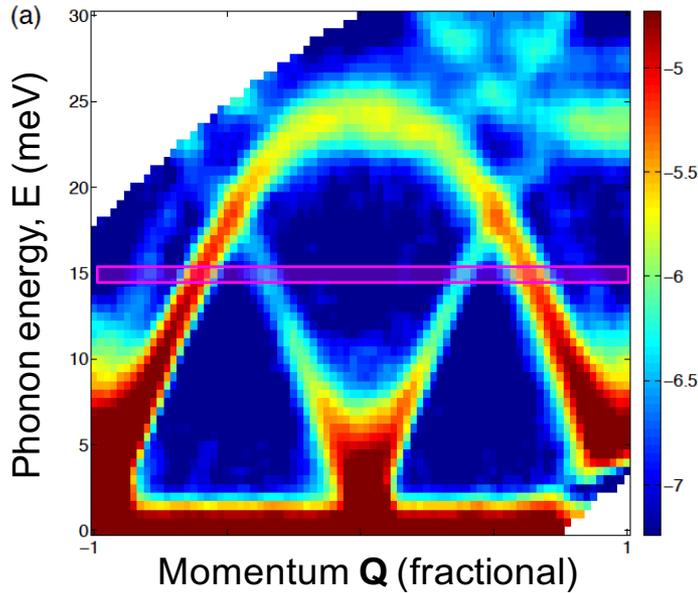
Matthiessen's rule:

$$\tau_{tot}^{-1} = \tau_{anh}^{-1} + \tau_{el-ph}^{-1} + \tau_{defect}^{-1} + \tau_{boundary}^{-1} + \dots$$

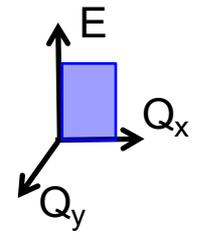
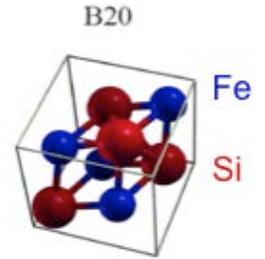
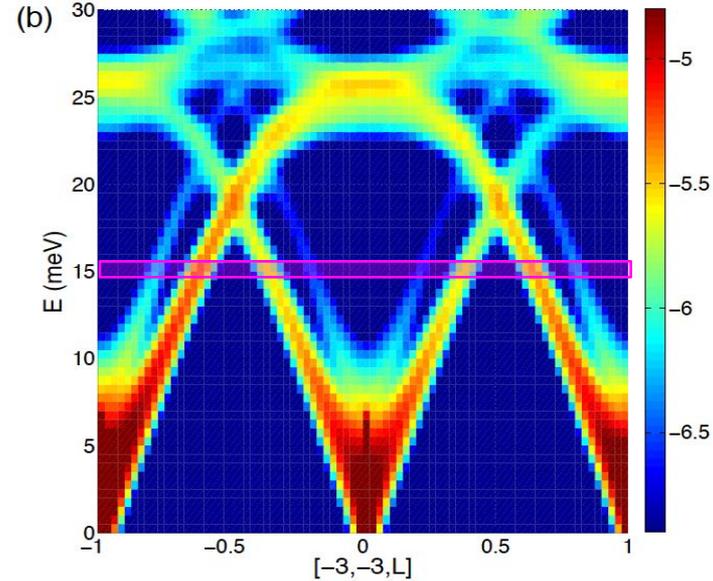
First-principles modeling of $S(\mathbf{Q}, E)$

Example: $S(\mathbf{Q}, E)$ in FeSi single-crystal

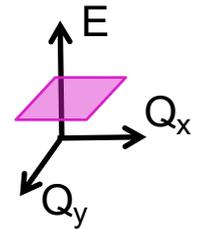
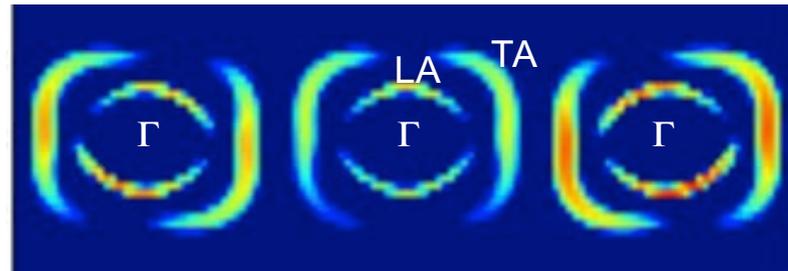
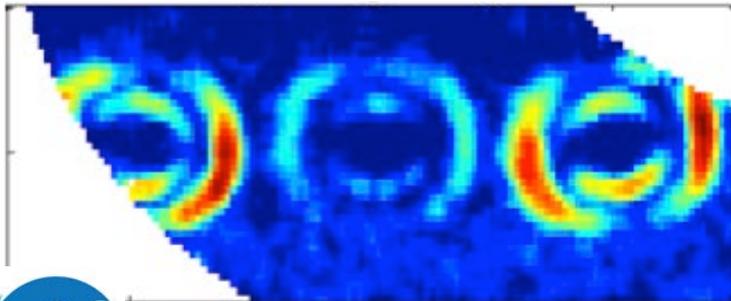
FeSi: neutron scattering



FeSi: DFT simulation



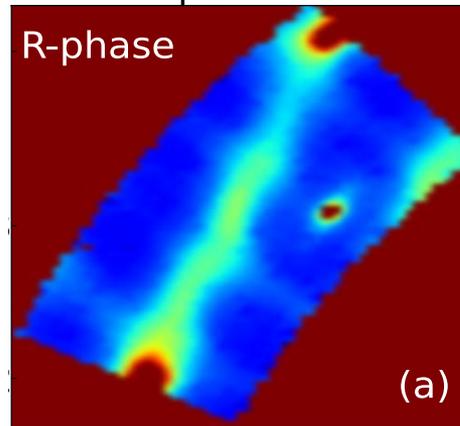
Measured intensities reflect phonon polarization vectors:



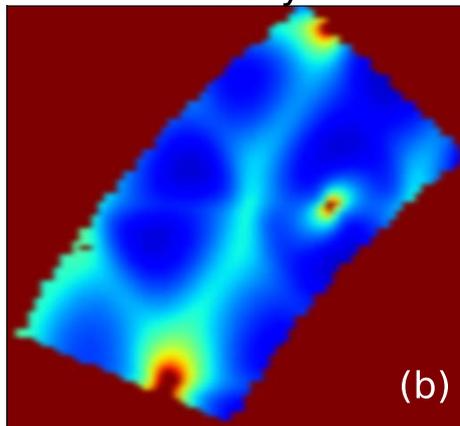
Thermal diffuse scattering DFT modeling

APS: VO₂ T-dependent

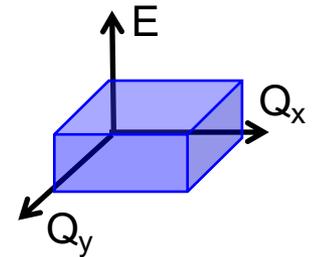
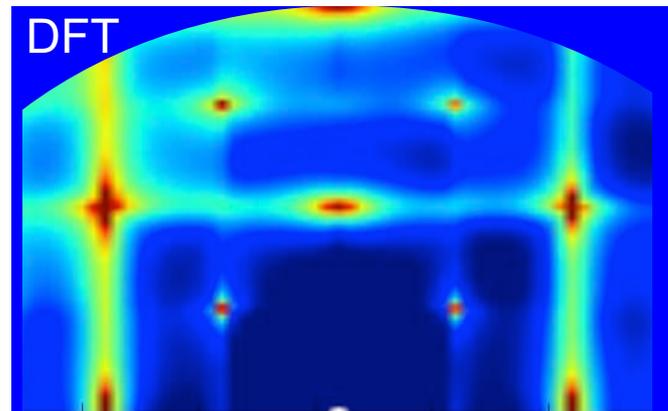
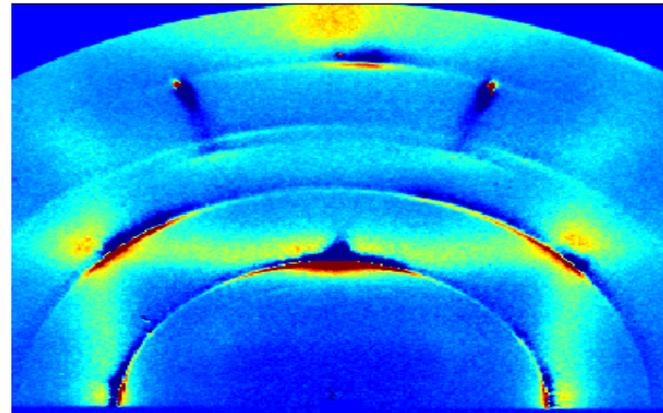
Experiment



Theory



PbTe, SSRL
with D. Reis' group, Stanford/SLAC

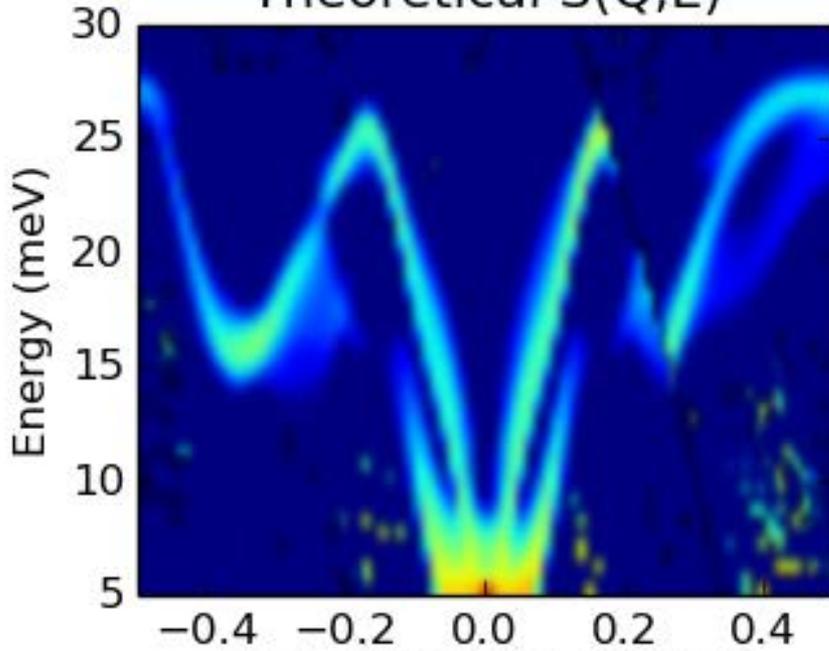


Budai*, Hong*, Manley, Specht, Li, Tischler, Abernathy, Said, Leu, Boatner, McQueeney, and Delaire, "Metallization of vanadium dioxide driven by large phonon entropy." Nature 515, 535 (2014)

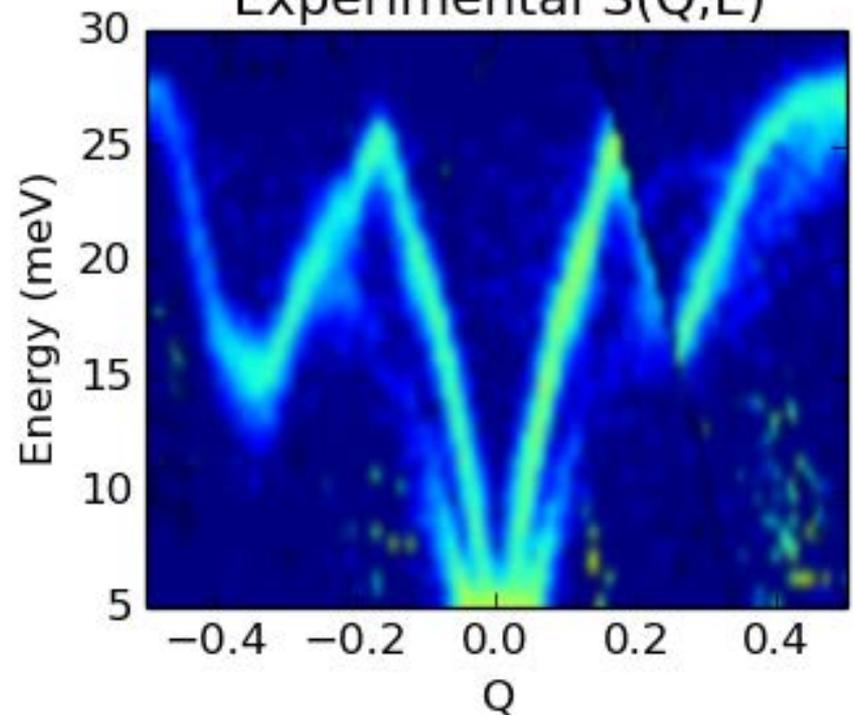
S(Q,E) Modeling: Comparison Between Experiment and Simulations: niobium

(DFT + optimization)

Theoretical S(Q,E)



Experimental S(Q,E)

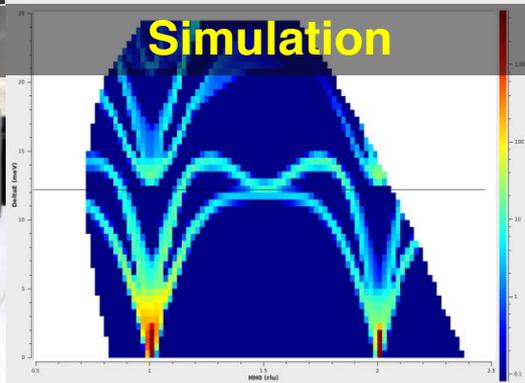
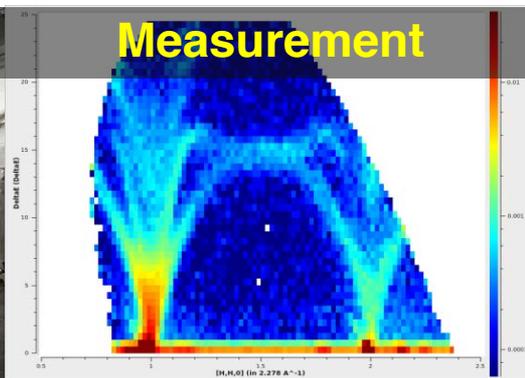


Bao, Archibald, Bansal, Delaire J. Comp. Physics, 315 (2016)

(our *Simphonies* software is available from GitHub)



CAMM supports concurrent inelastic neutron scattering measurements and first-principles modeling of anharmonic phonons



- Study of ferroelectric instabilities in SrTiO_3
- Measurements on HYSPEC using live data streaming (top)
- Dedicated access to Cray XC30 EOS cluster at Oak Ridge Leadership Computing Facility (11,000 cores)
- Full scale ab-initio molecular dynamics simulations on experiment timescale allowing real time decisions

O. Delaire, J. Hong, H. Cao, A. Savici, B. Winn, L. Boatner, G. Shipman, manuscript in preparation

MANTiD



<http://camm.ornl.gov>

Effort supported by Center for Accelerating Materials Modeling.

PbTe: strong phonon anharmonicity near ferroelectric instability

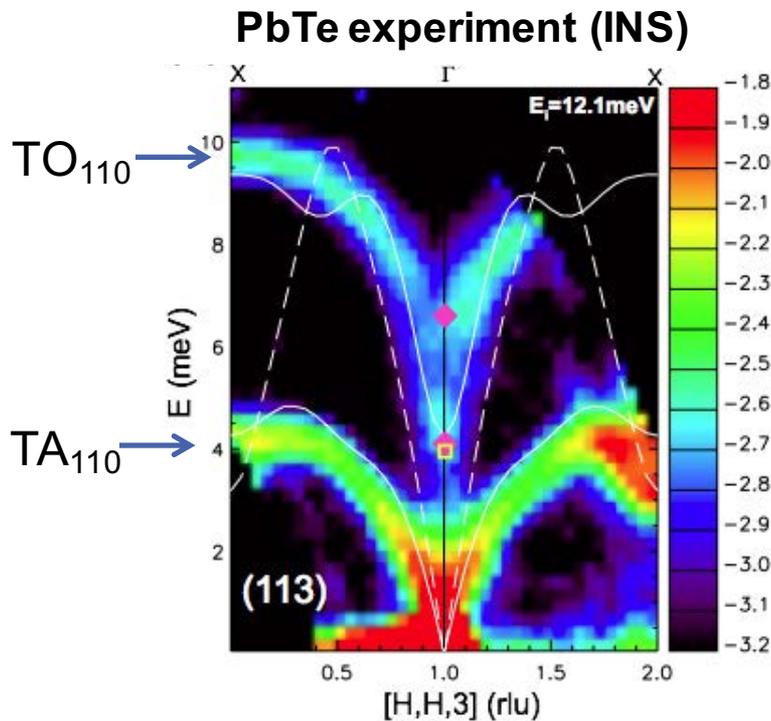
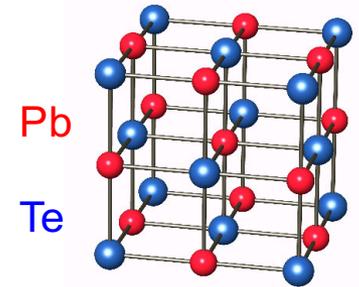
Delaire, Ma *et al.*, ***Nature Materials*** 10, 614 (2011)

Li, Hellman, Delaire *et al.*, ***Phys. Rev. Lett.*** 112, 175501 (2014)

Li, Ma, Delaire *et al.* ***Phys. Rev. B*** 90, 214303 (2014)

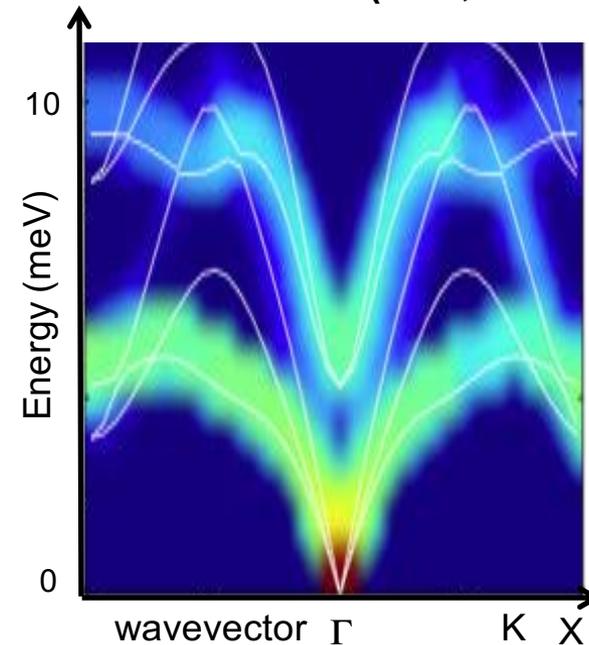
Anharmonicity in rocksalt PbTe

- Large figure-of-merit $ZT_{\max} \sim 2$ (Pei Nature 2011, Biswas Nature 2012)
- Low thermal conductivity $\kappa_{\text{lat}} = 2 \text{ W/m/K}$ at 300K
- **Strongly anharmonic transverse-optic (TO) mode (“soft mode”).**
- **5x suppression in κ_{lat} from *optic-acoustic scattering*** (Shiga PRB 2012, Tian PRB 2012)



CNCS @ SNS (12meV)

PbTe calculation (DFT, harmonic)



Delaire, Ma *et al.*, Nature Materials 2011

Time-resolved thermal diffuse scattering: LCLS measurements on PbTe

Photoinduced suppression of the ferroelectric instability in PbTe

Jiang *et al.* Nature Commun. 2016 (collaboration with D. Reis)

- No evidence for off-centering/dipoles in diffuse scattering

>>> Does not support conclusions of Bozin, Billinge *et al.*, “Entropically Stabilized Local Dipole Formation in Lead Chalcogenides”, Science (2010)

- Diffuse scattering entirely from phonons (THz timescale)

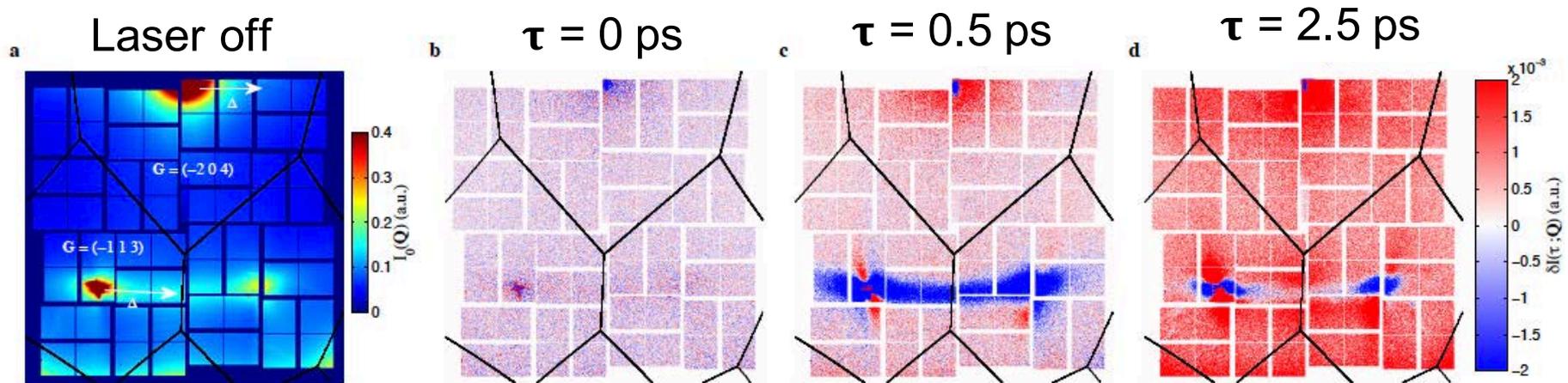


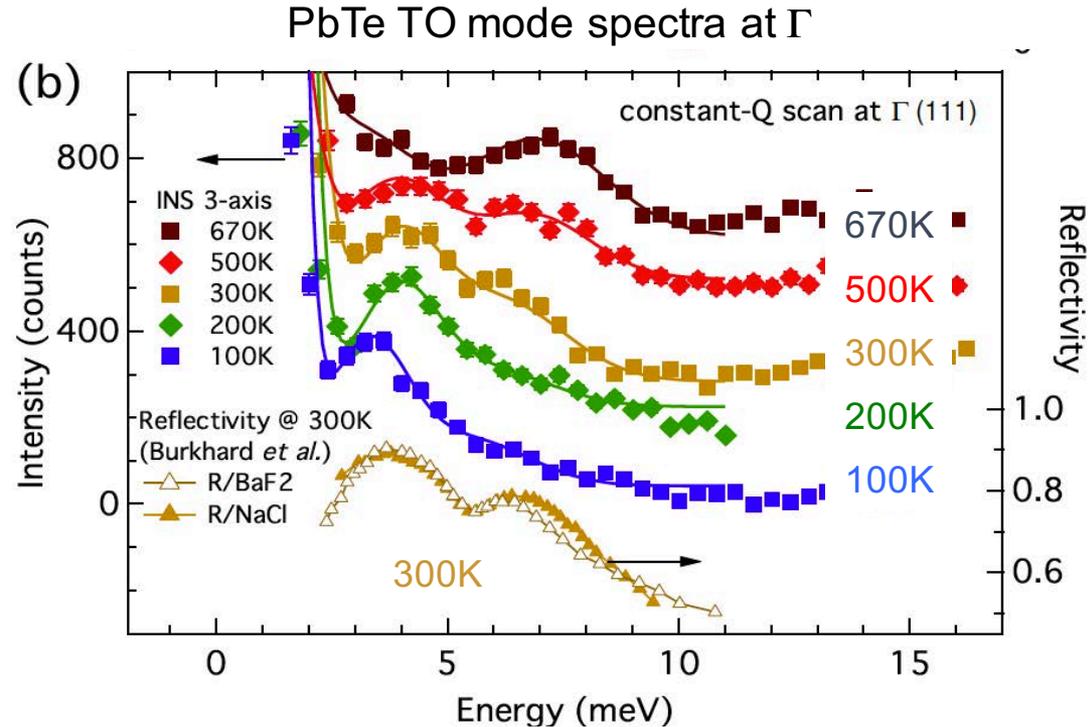
Figure 1 | Femtosecond x-ray diffuse scattering from PbTe. a, Reference scattering from PbTe prior to photoexcitation.

PbTe single-crystal time-resolved pump-probe TDS on LCLS / XPP

T-dependence of TO phonon in PbTe

Deviates strongly from quasi-harmonic behavior:

- Broad TO (ferroelectric) mode at Γ , stiffens with increasing T.
- Double-peak structure, with strong T-dependence.



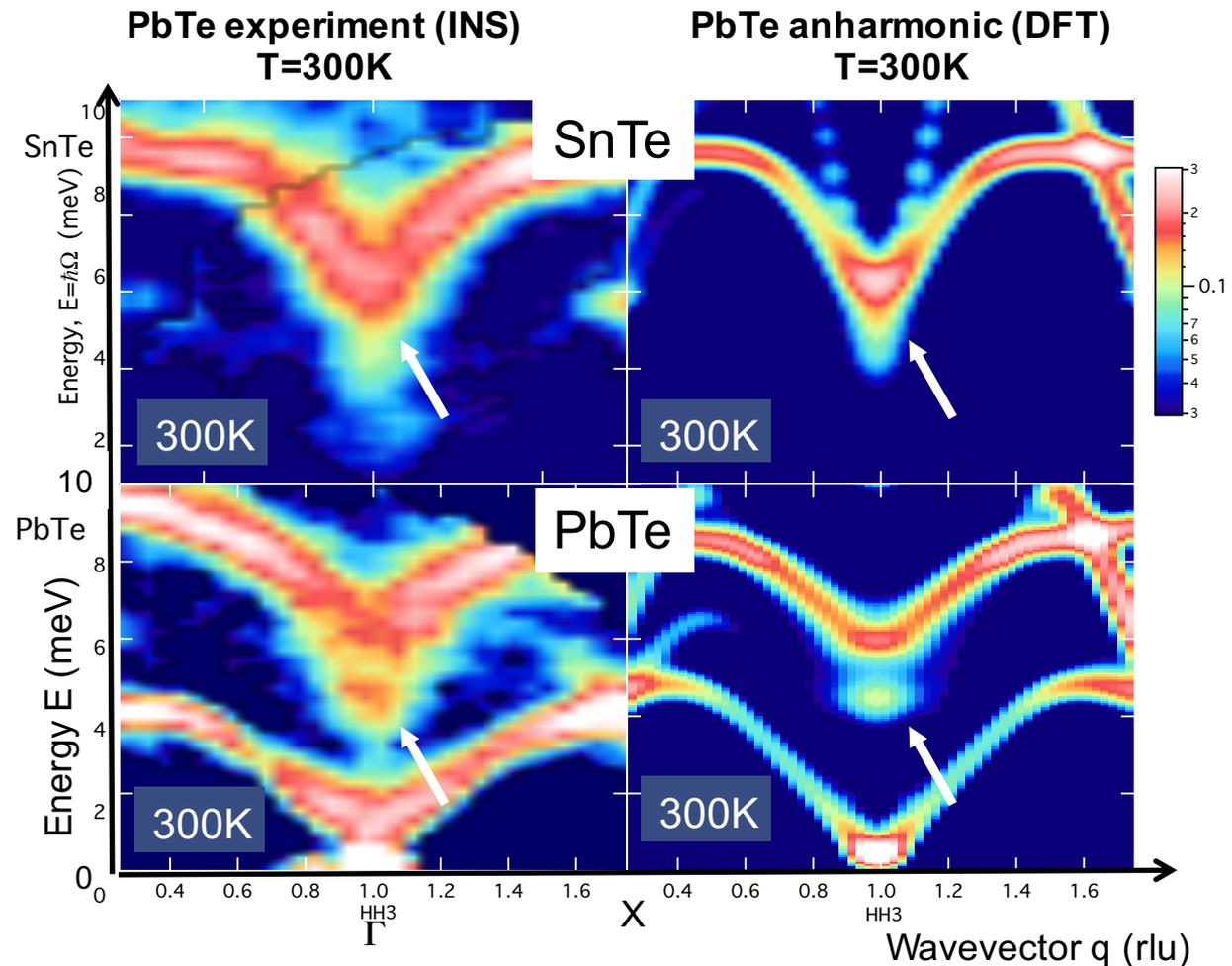
HB3 @ HFIR (thermal 3-axis)

Delaire, Ma *et al.*, Nature Materials **10**, 614 (2011)

Confirmed by Jensen *et al.* PRB **86**, 085313 (2012)

Anharmonic phonon spectral functions

- **Ab-initio MD + TDEP reproduces strong anharmonic effects**
- $\chi''(Q,E)$ computed from anharmonic phonon self-energy $\Sigma_q(E)$
- Reproduce TO phonon splitting in PbTe, by including anharmonicity



Phys. Rev. Letters 112, 175501 (2014)
Phys. Rev. B 90, 214303 (2014)

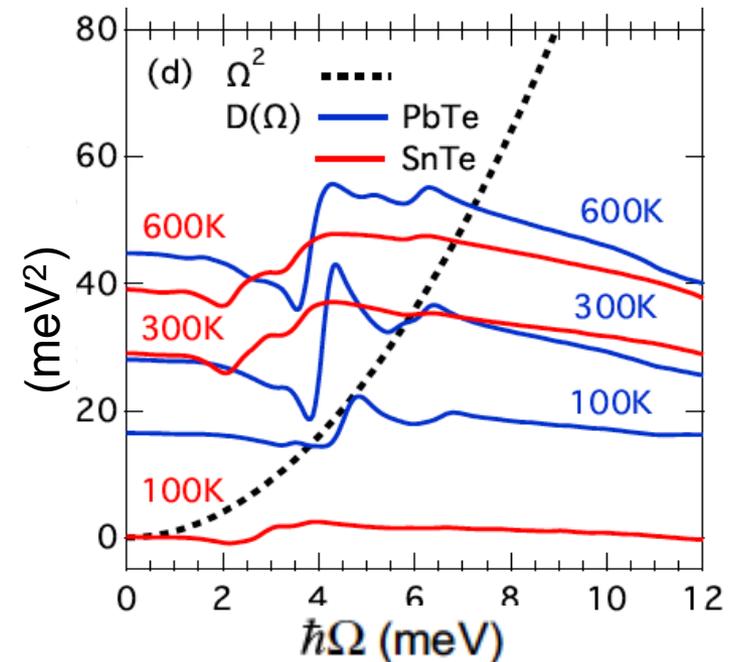
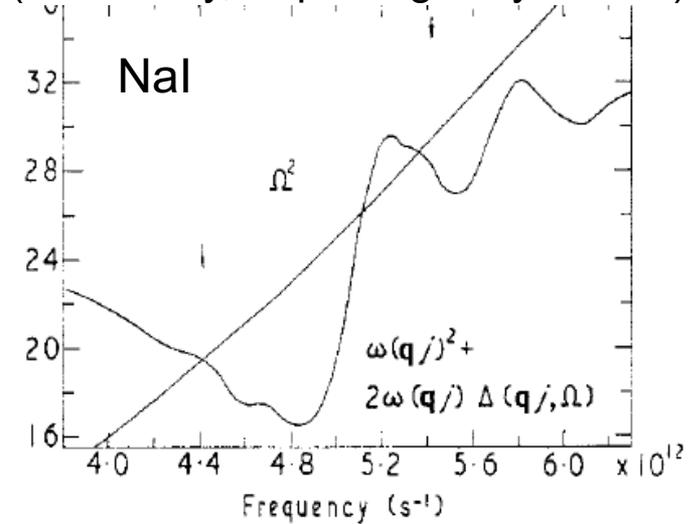
Neutron spectra as probes of many-body effects

- Dynamical susceptibility:

$$\chi''_{\mathbf{q}j}(\Omega) = |F(\mathbf{Q}, \Omega)|^2 \times \frac{2\omega_{\mathbf{q}j}\Gamma_{\mathbf{q}j}(\Omega)}{\{\Omega^2 - \omega_{\mathbf{q}j}^2 - 2\omega_{\mathbf{q}j}\Delta_{\mathbf{q}j}(\Omega)\}^2 + 4\omega_{\mathbf{q}j}^2\Gamma_{\mathbf{q}j}(\Omega)^2}$$

- Look for minima in {...}
- Multiple crossings lead to multiple peaks in the INS spectra for $\chi''_{\mathbf{q}}(E)$
- $\Delta_{\mathbf{q},j}(\Omega)$ obtained from anharmonic interatomic force-constants.

(R. Cowley, Rep. Prog. Phys. 1968)



Phonon “nesting” increases phase-space for acoustic-optic scattering of TO mode

Imaginary part of self-energy (damping):

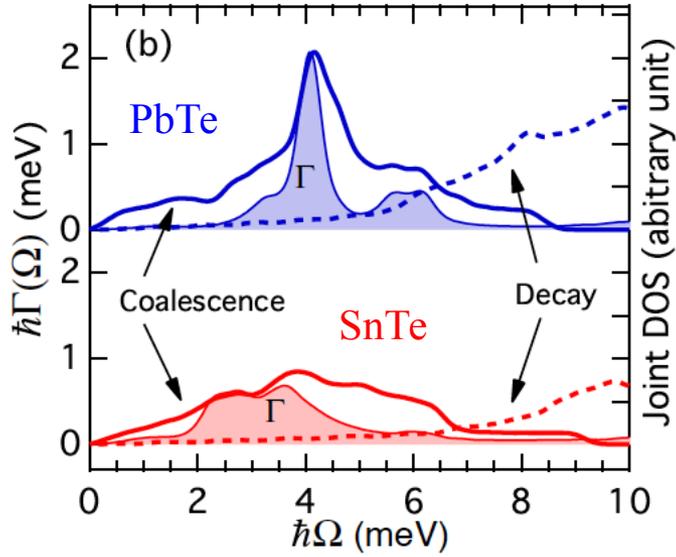
$$\Gamma_{\mathbf{q}j}(\Omega) = \frac{18}{\hbar^2} \sum_{\mathbf{q}_1 \mathbf{q}_2 j_1 j_2} |V_3 \begin{pmatrix} \mathbf{q} & \mathbf{q}_1 & \mathbf{q}_2 \\ j & j_1 & j_2 \end{pmatrix}|^2 \times$$

$$\left((n_1 + n_2 + 1) [\delta_{\omega_1 + \omega_2 - \Omega} - \delta_{\omega_1 + \omega_2 + \Omega}] + (n_2 - n_1) [\delta_{\omega_1 - \omega_2 - \Omega} - \delta_{\omega_1 - \omega_2 + \Omega}] \right),$$

3-phonon interaction strength
(transition probability)

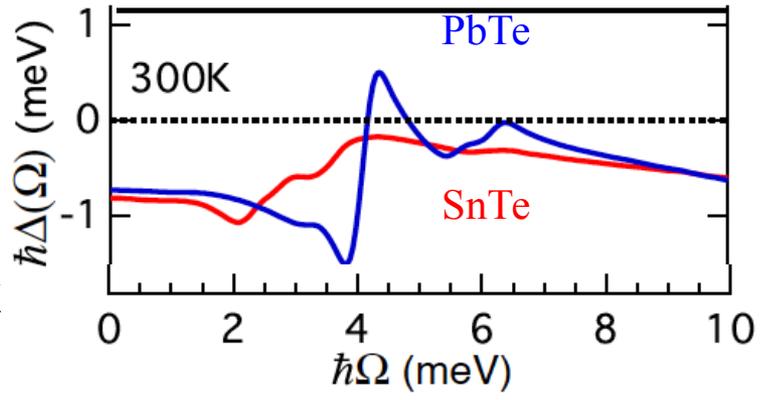
Kinematic constraints for conserving energy and momentum
(phase space size ~ joint DOS)

Real and imaginary parts of self-energy related by Kramers-Kronig (Hilbert) transformation:



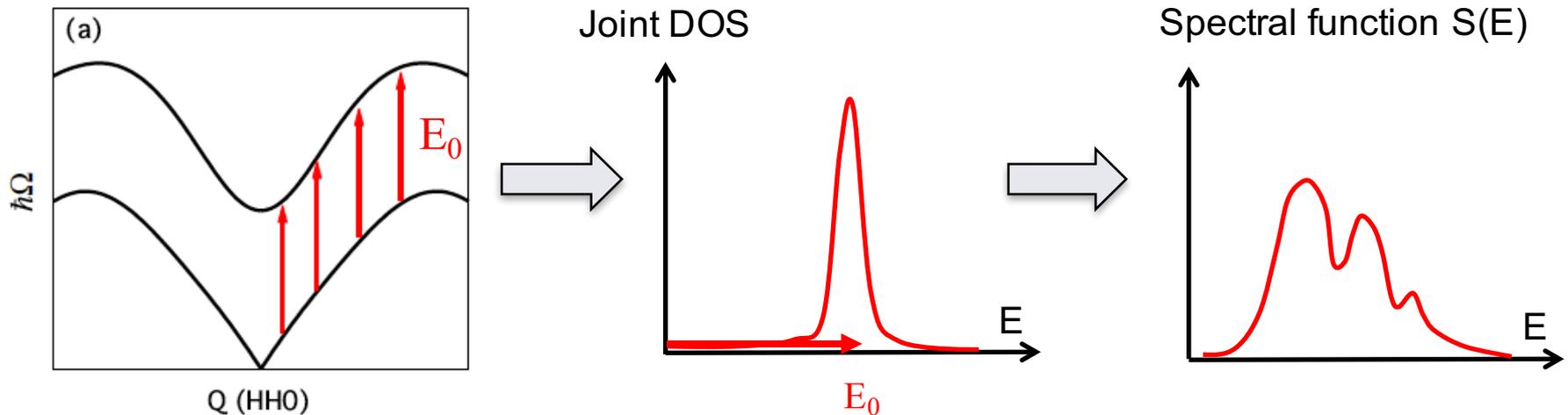
$$\mathcal{H}$$

$$\mathcal{H}[\delta(x)] = 1/x$$



Phonon “nesting” increases phase-space for acoustic-optic scattering of TO mode

- Phase-space size amplifies effect of anharmonic potential
- Nesting of phonon dispersions:



→ *screen/engineer materials for favorable dispersions*

SnSe: bonding instability, phase transition and anharmonicity

Li*, Hong*, May, Bansal, Ma, Hong, Chi, Ehlers, and Delaire, *Nature Physics* 11, 1063 (2015)

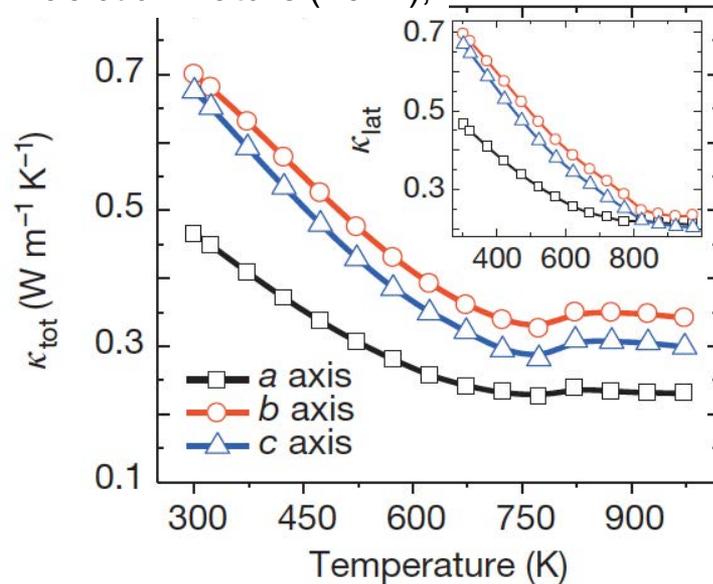
Bansal, Hong, Li, May, and Delaire *Phys. Rev. B* 94, 054307 (2016)

Hong and Delaire, arxiv:1604.07077

SnSe: anisotropy and phase transition

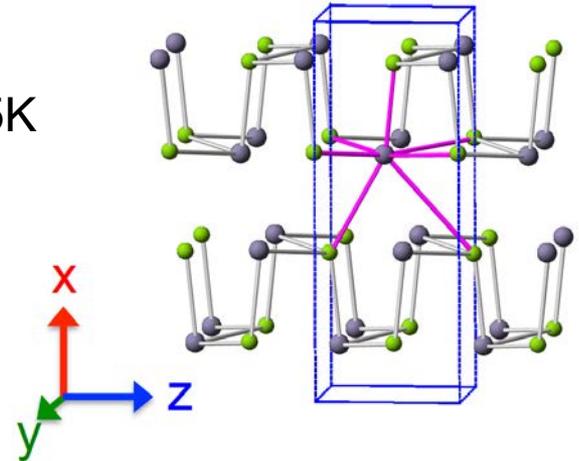
- Very high $zT \sim 2.5$ and very low thermal conductivity $\kappa_{\text{lat}} < 1 \text{ Wm}^{-1}\text{K}^{-1}$
- Strongly anisotropic,
- Structural phase transition at $\sim 805\text{K}$

Zhao *et al.* Nature (2014),

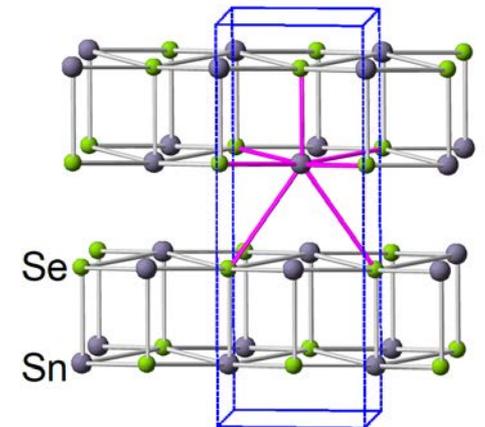


Zhao *et al.* Nature (2014), Zhao *et al.* Science (2016),
Chen *et al.* J. Mater. Chem. 2014, Carrete *et al.* APL (2014)

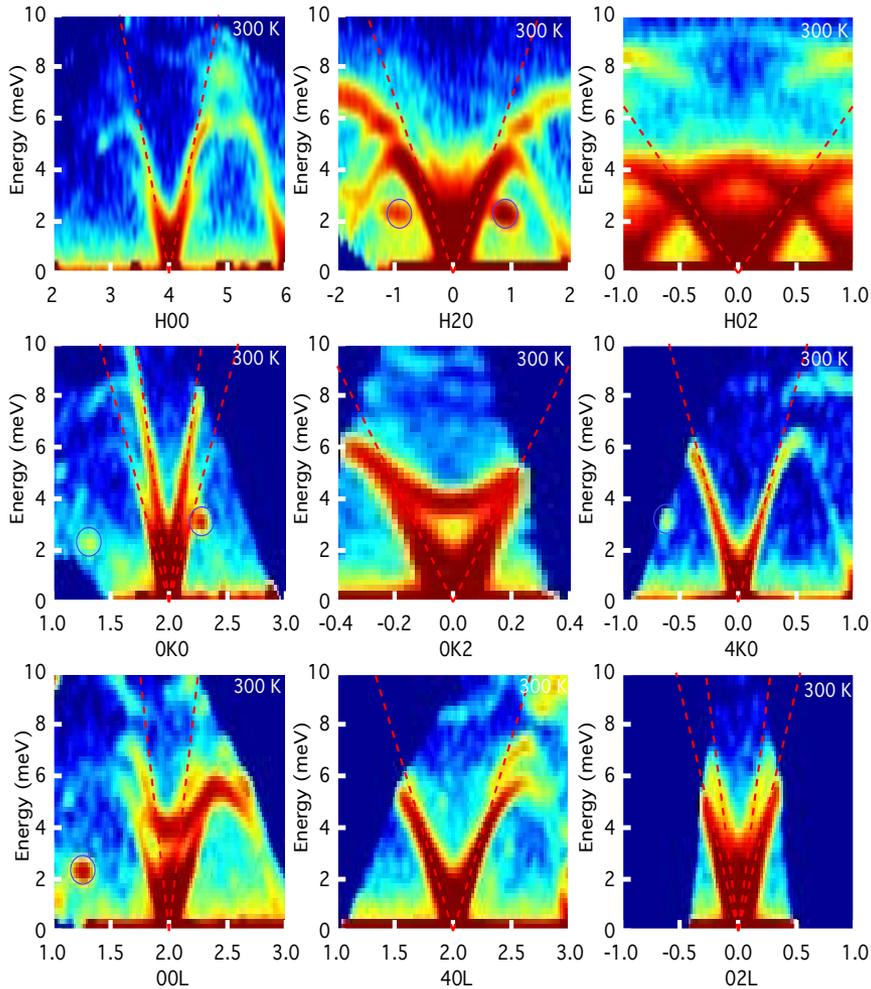
Pnma
 $T < 805\text{K}$



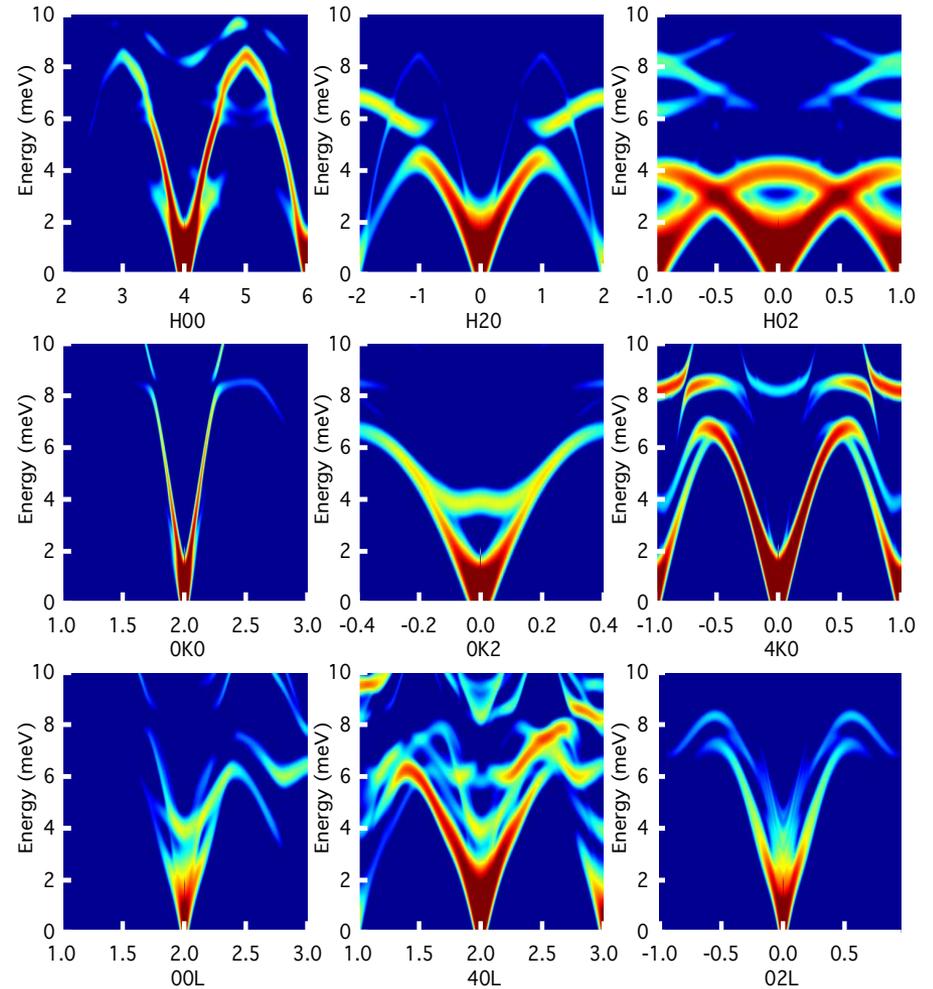
Cmcm
 $T > 805\text{K}$



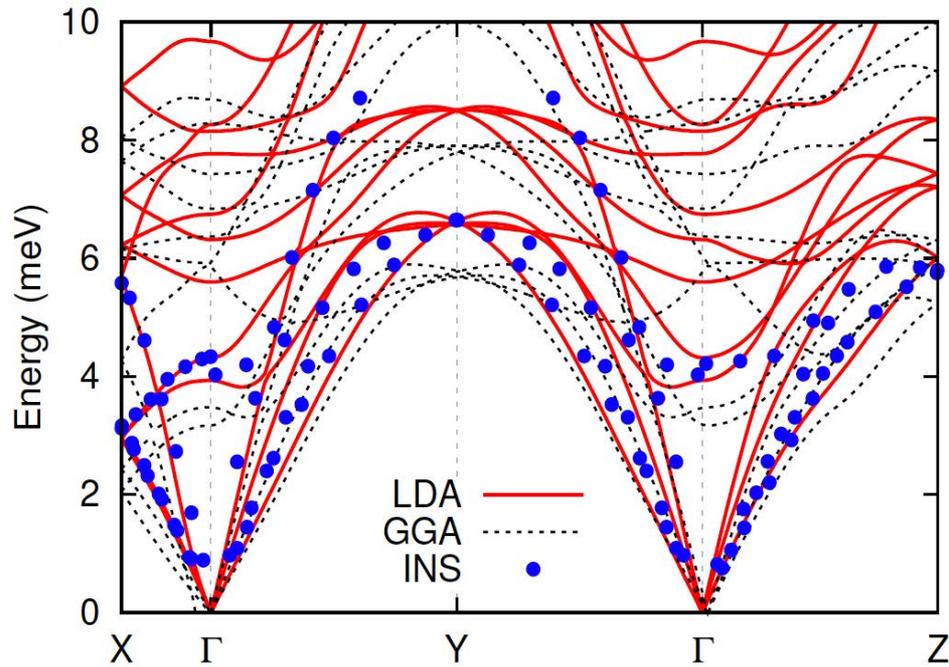
Neutron scattering



First-principles simulations



***Important to preserve information
about the polarization vectors***



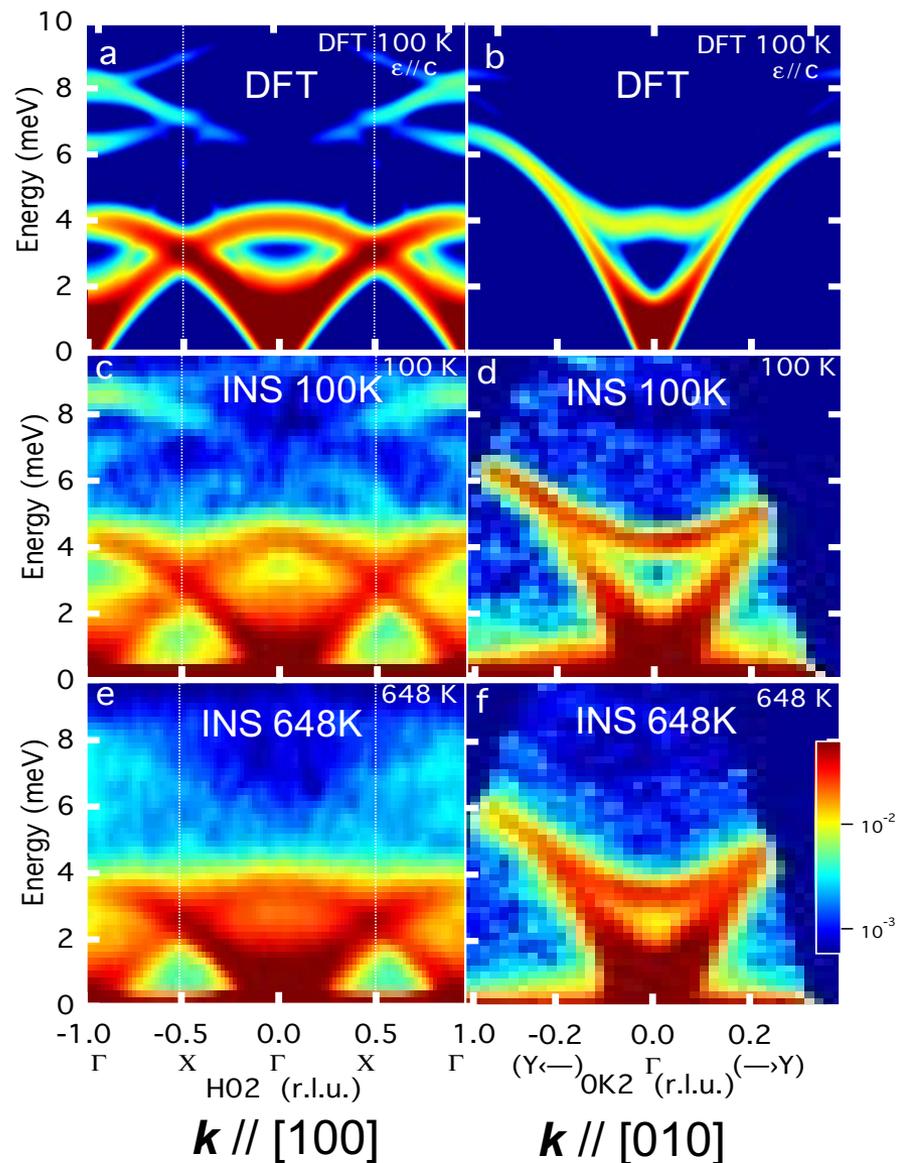
LDA more accurate than GGA (PBE)

FIG. 2. **GGA (PBE)** ons
 with either LDA or GGA exchange-correlation functionals (re-
 relaxed unit cells), compared with experimental INS data at
 300 K.

SnSe: $S(Q,E)$ for *c*-polarized modes

- Measured phonons with neutron scattering on single-crystals at $100\text{K} < T < 850\text{K}$
- Mapped all crystallographic directions (orthorhombic $a \neq b \neq c$)
- Observe strong softening with increasing temperature (up to $Pnma - Cmcm$ phase transition)
- Especially anharmonic modes with polarizations along *c*-axis.
- Good agreement with DFT / LDA but GGA underestimates phonon frequencies.

Li*, Hong*, May, Bansal, Ma, Hong, Chi, Ehlers, and Delaire, Nature Physics 11, 1063 (2015)

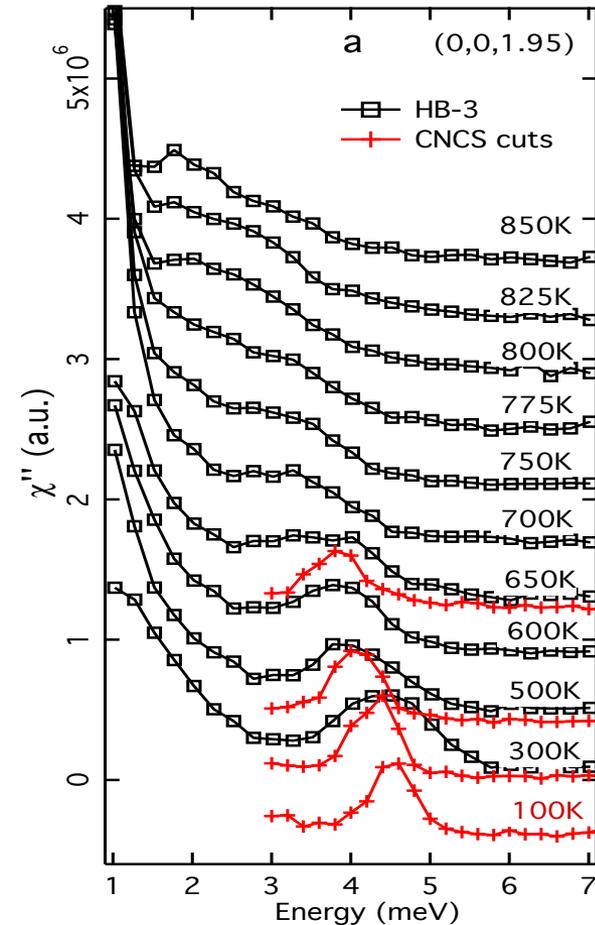


CNCS @ SNS (12meV)

SnSe soft-mode behavior of $\text{TO}_c(\Gamma)$

- Many acoustic and optical phonons show strong temperature dependence.
- Lowest zone-center TO_c mode softens at the phase transition.
- Confirms strong anharmonicity.

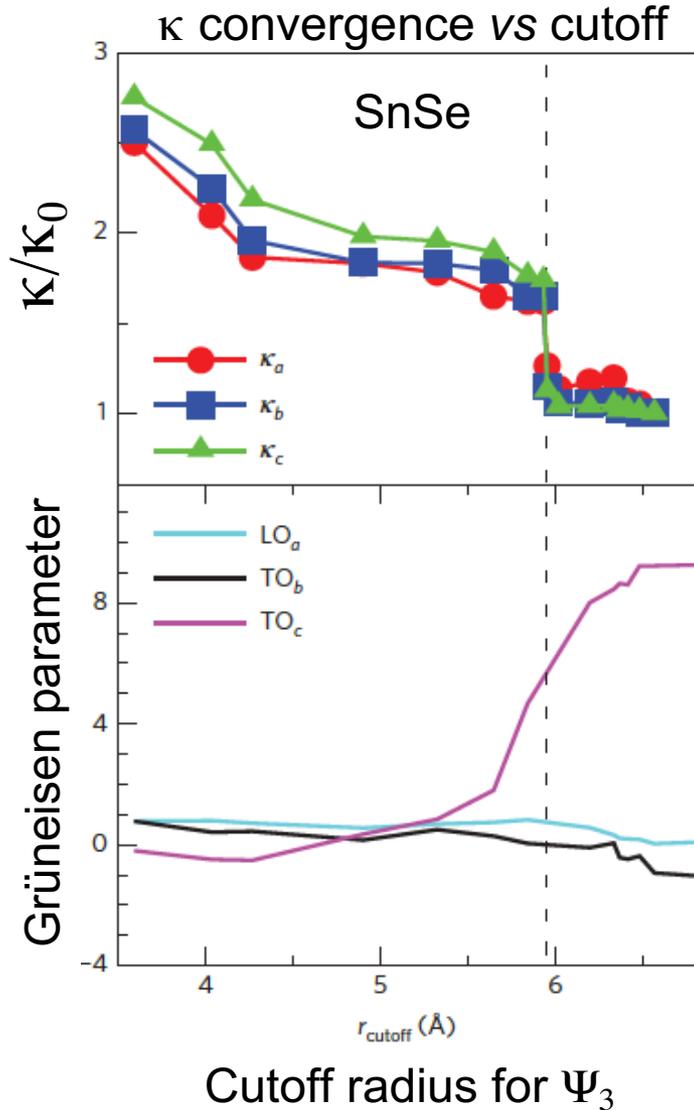
SnSe zone-center TO_c



Li*, Hong*, May, Bansal, Ma, Hong, Chi, Ehlers, and Delaire, Nature Physics 11, 1063 (2015)

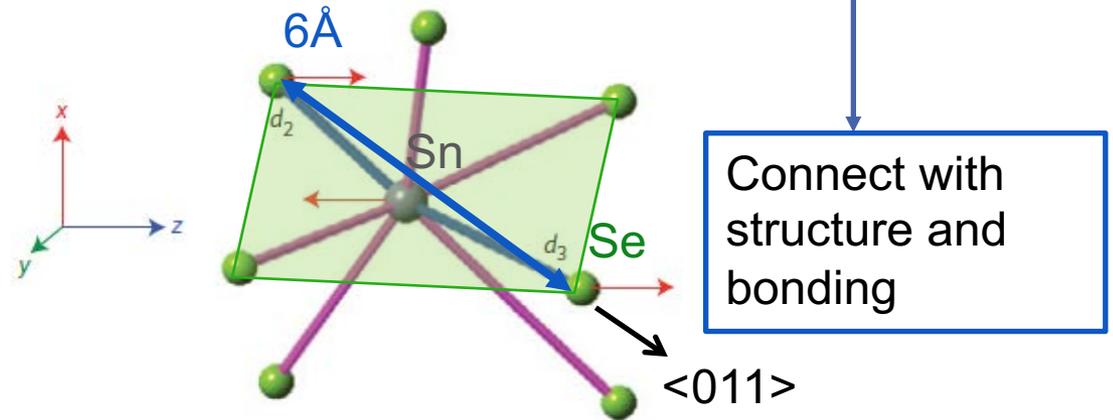
HB3 @ HFIR
CNCS @ SNS

Anharmonicity and thermal transport



- SnSe thermal conductivity computed from first-principles.
- Strong influence of cubic terms Ψ within Sn 1NN coordination polyhedron for triplet with Se-Se distance $\sim 6\text{Å}$

$$V = V_0 + \frac{1}{2!} \sum_{ij,\alpha,\beta} \Phi_{ij}^{\alpha,\beta} u_i^\alpha u_j^\beta + \frac{1}{3!} \sum_{ij,k,\alpha,\beta,\gamma} \Psi_{ij,k}^{\alpha,\beta,\gamma} u_i^\alpha u_j^\beta u_k^\gamma + \dots$$



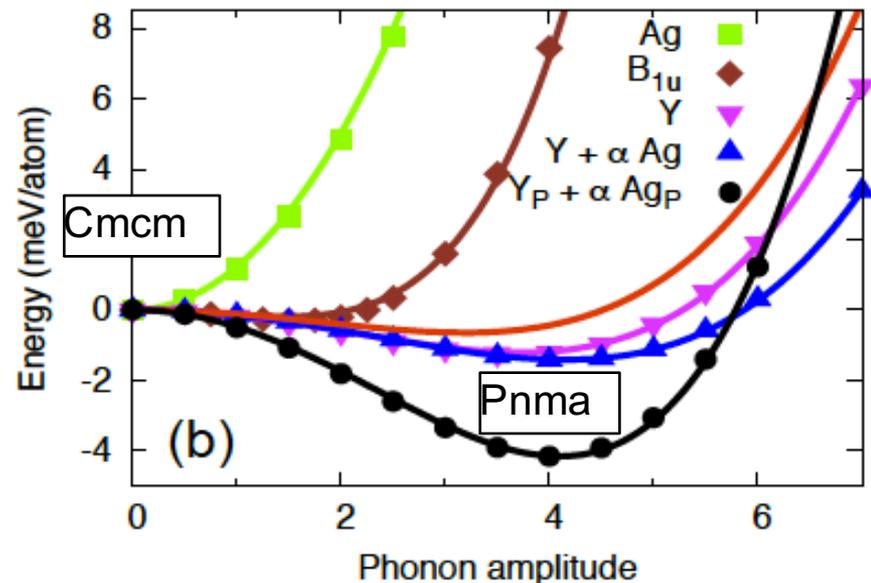
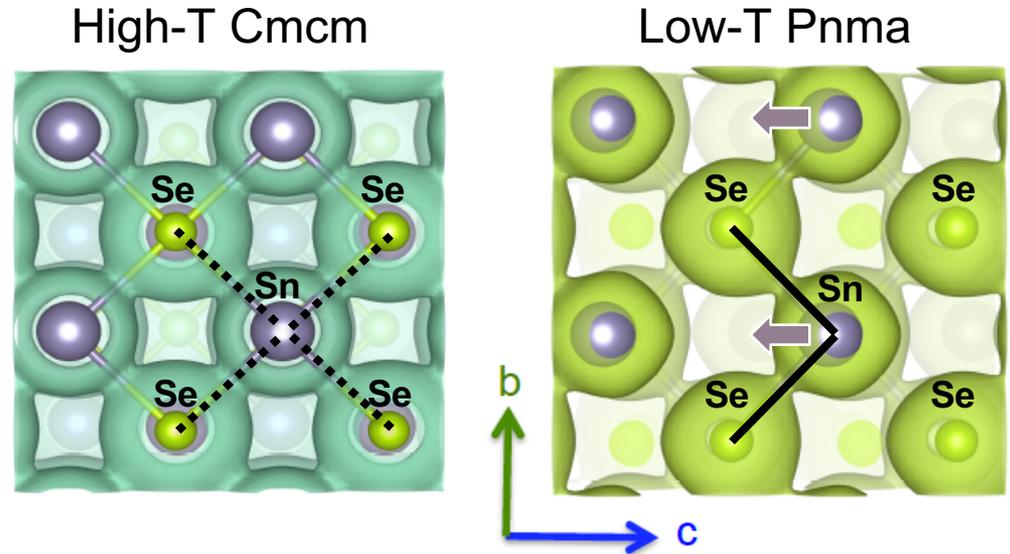
Li*, Hong*, May, Bansal, Ma, Hong, Chi, Ehlers, and Delaire, Nature Physics 11, 1063 (2015)

*W. Li Comput. Phys. Commun. 185, 17471758 (2014)

Anharmonicity from bonding instability

Electron Localization Function

- Electronic instability (Jahn-Teller) of high-T Cmcm phase: Resonantly-bonded high-symmetry Cmcm phase distorts to Pnma to lower electronic energy.
- Results in double-well anharmonic potential for ions.
- **We have identified several strongly anharmonic materials near Jahn-Teller / Peierls instabilities.**



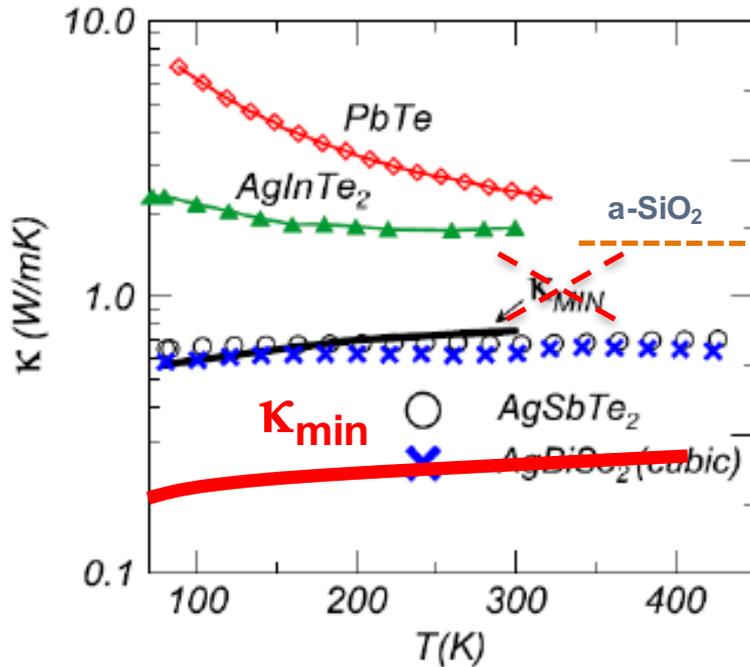
Hong and Delaire,
arxiv:1604.07077

AgSbTe₂ : importance of nanostructure

J. Ma*, O. Delaire*, *et al.*, ***Nature Nanotechnology*** 8, 445 (2013)

Ma, Delaire *et al.* ***Phys. Rev. B*** 90, 134303 (2014)

AgSbTe₂ vs PbTe: the importance of nanostructure



Morelli *et al.* PRL 2008

Why are the thermal conductivities of PbTe and AgSbTe₂ so different?

“Ioffe-Regel limit”

= *minimum* lattice thermal conductivity

(mean-free-paths of phonons cannot be smaller than interatomic distance)

Ma*, Delaire* *et al.* Nature Nanotechnology 2013

AgSbTe₂: glass-like κ_{lat} , but not minimum

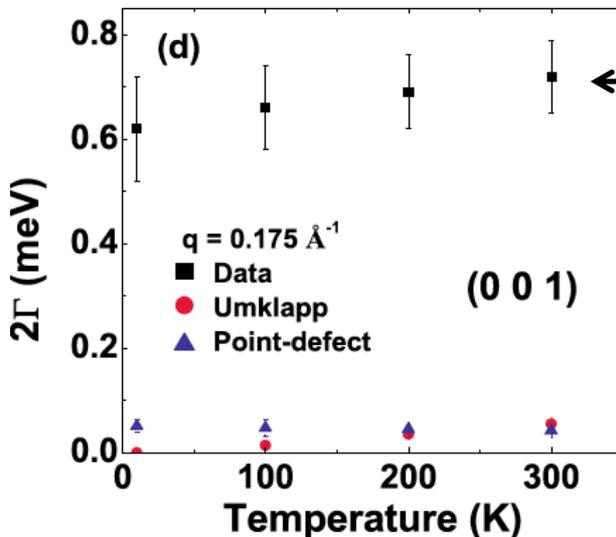
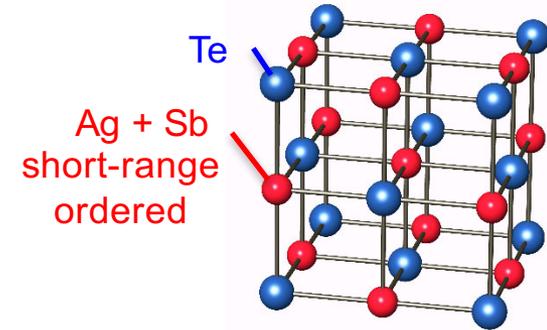
- Estimate thermal conductivity from INS data:

Transport: $\kappa_{\text{lat}} = 0.7 \text{ Wm}^{-1}\text{K}^{-1}$ at 300K

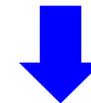
From INS: $\kappa_{\text{lat}} = 0.8 \pm 0.15 \text{ Wm}^{-1}\text{K}^{-1}$ at 300K

- INS shows broad linewidths (but not fully damped), independent of temperature: structural scattering

→ INS provides *microscopic view on thermal transport*.



Linewidth independent of T (INS)



Anharmonicity **not** dominant for $T < 300\text{K}$
(\neq Morrelli's conjecture)

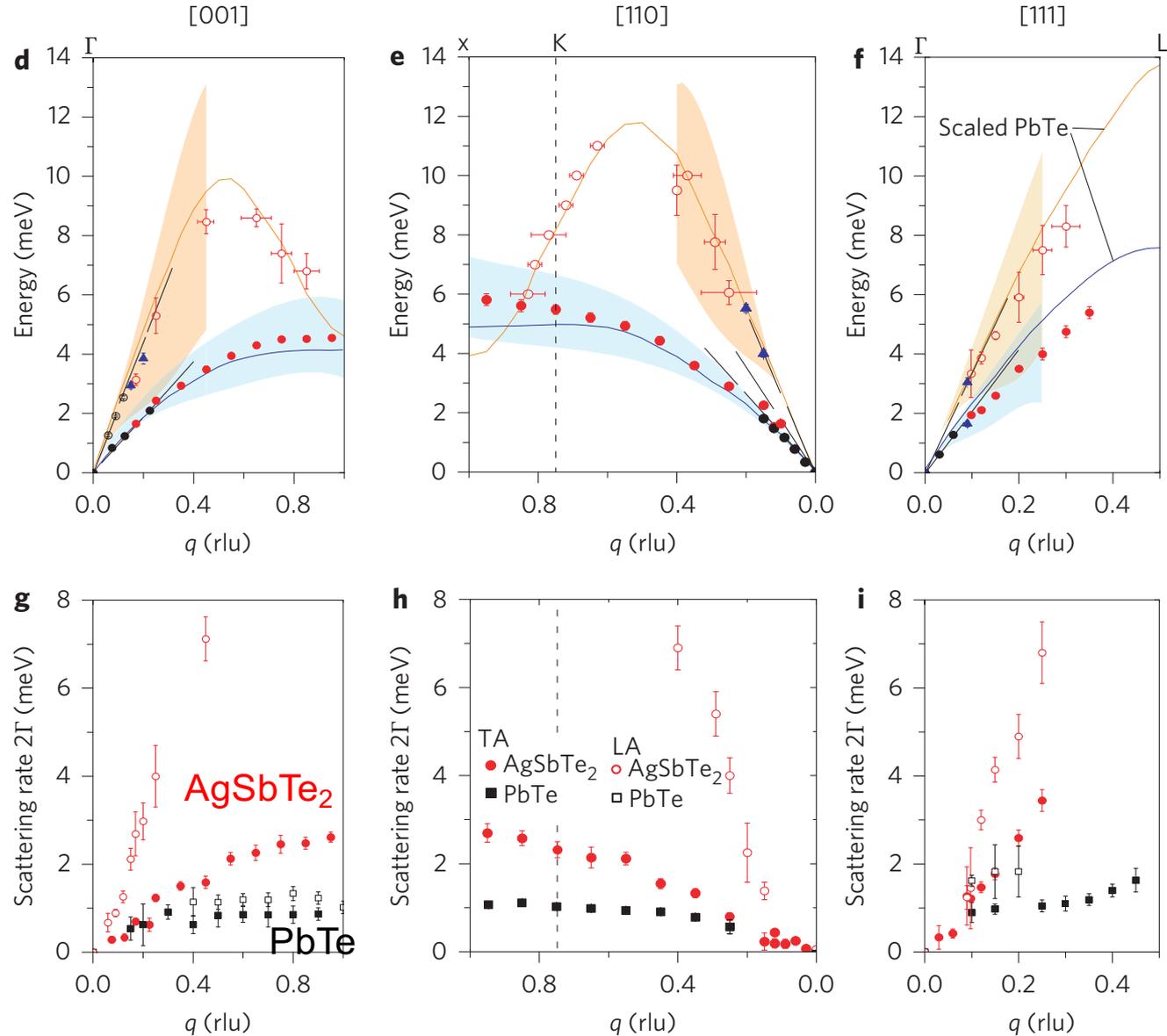
Ma*, Delaire* *et al.* Nature Nanotechnology 2013

AgSbTe₂ phonon dispersions and lifetimes

Mapped dispersions
in AgSbTe₂ (INS):

dispersions similar to
PbTe (scaled by $M^{1/2}$).

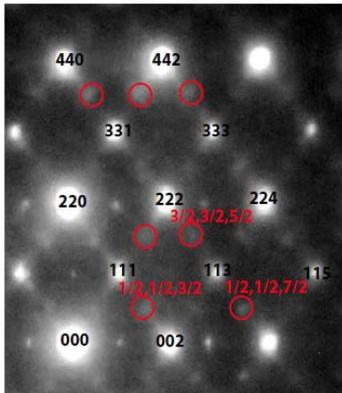
**Scattering rates
are much larger in
AgSbTe₂ than in
PbTe.**



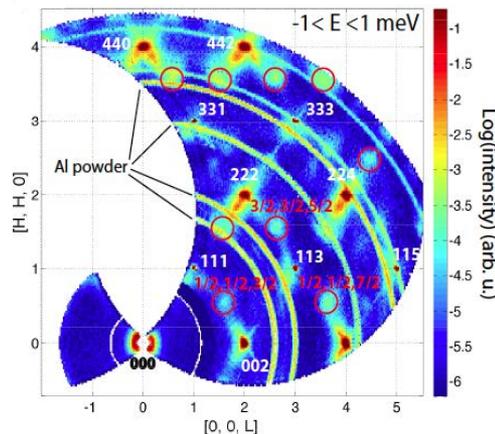
Diffraction reveals nano-scale ordering

- Short-range ordering of Ag/Sb seen with neutrons, x-rays, and TEM, agreement with DFT prediction (Barabash PRL 2008).
- Superlattice half-integer peaks (cation ordering), rods along $\langle 111 \rangle$, and lobes.
- Ordering correlation length $\xi \sim 3\text{nm}$. Nanoscale ordering twins distort lattice.

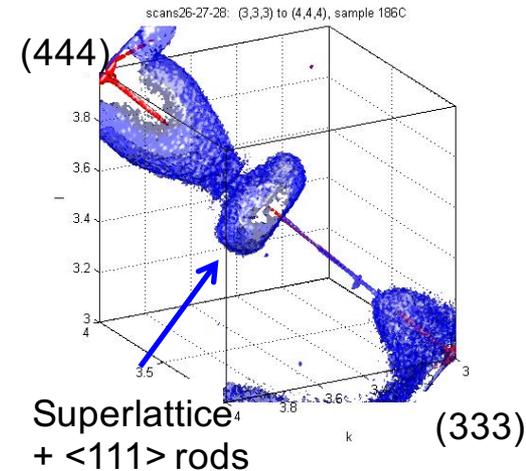
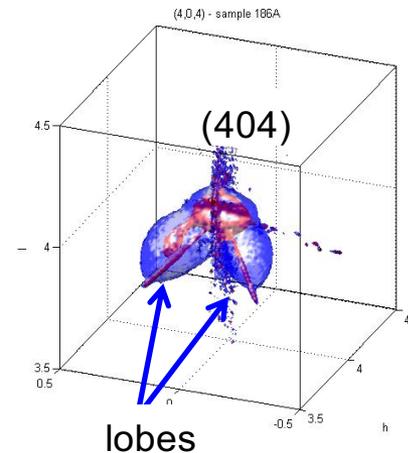
TEM diffraction:
diffuse + superlattice
(1,-1,0) reciprocal plane



Neutron elastic scattering:
diffuse + superlattice
(1,-1,0) reciprocal plane



X-ray diffuse scattering (APS 33BM):
superlattice + lobes + $\langle 111 \rangle$ rods

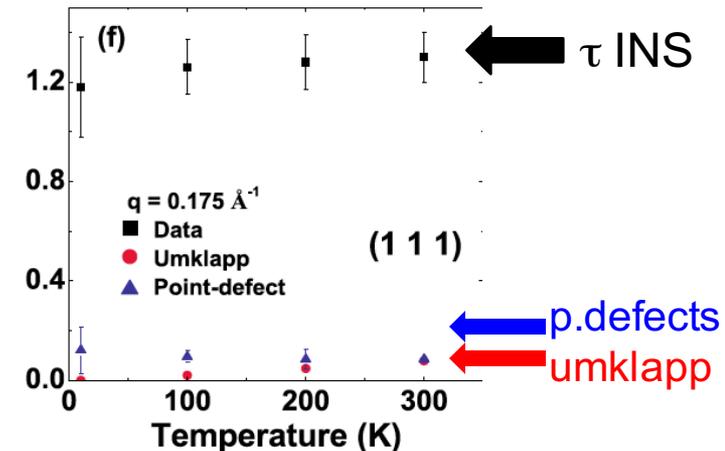
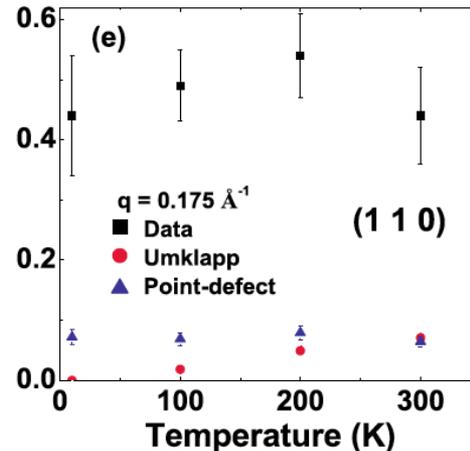
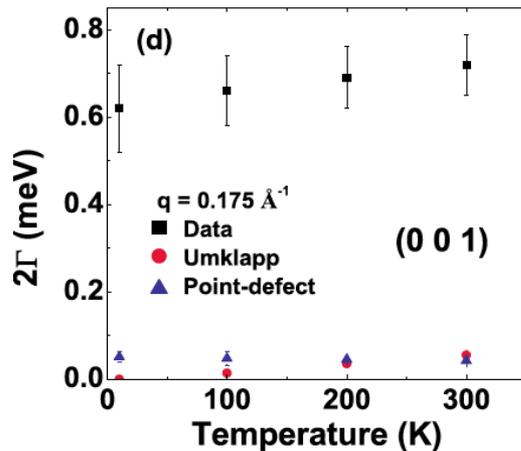


Ma, Delaire *et al.* Nature Nano 2013

Ma, Delaire *et al.* PRB 2014

Temperature dependence of linewidths

- Phonon linewidths (scattering rates) **constant** with temperature \neq anharmonic.
- Measured widths much larger than umklapp or point-defect contributions.



Umklapp

$$\tau_u^{-1} \approx \frac{\hbar\gamma^2}{Mv^2\theta_D} \omega^2 T \exp\left(-\frac{\theta_D}{3T}\right)$$

With measured ω , θ_D , v , and $\gamma=2.1$

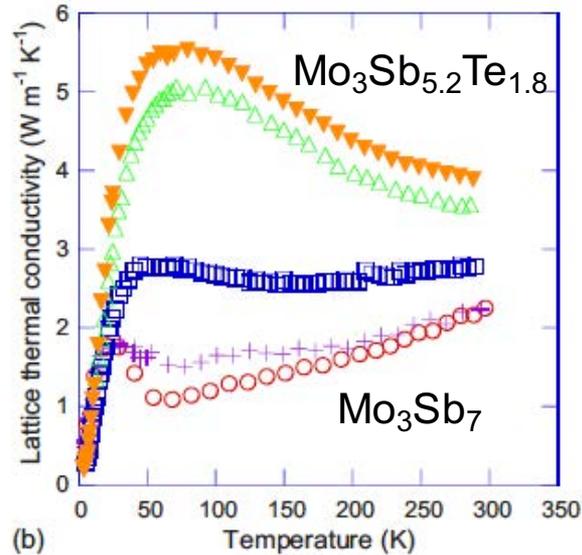
Point-defects

$$\tau_{p.d.}^{-1} = \frac{V}{4\pi v^3} \omega^4 \sum_i f_i \left[\frac{\bar{m} - m_i}{\bar{m}} + \frac{2(\bar{k} - k_i)}{\bar{k}} \right]^2$$

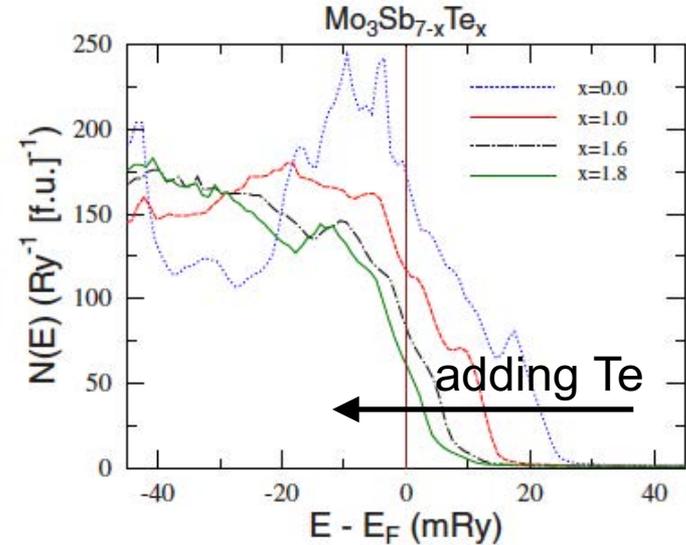
With calculated $k_{Sb}/k_{Ag} \sim 2$
And 5% vacancies

Electron-phonon coupling in $\text{Mo}_3\text{Sb}_{7-x}\text{Te}_x$

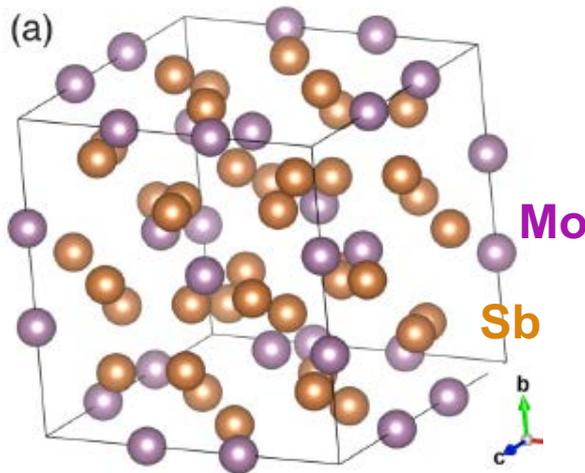
Lattice thermal conductivity vs Te-alloying



Valence band filling with Te substitution



From: Candolfi *et al.* Phys. Rev. B 79, 235108 2009



20 atoms in conventional cubic cell

Structural transition (cubic \rightarrow tetra) below 53K

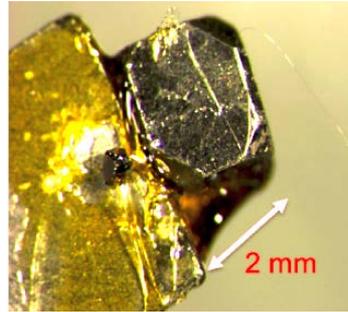
Electron-phonon coupling and thermal transport in the thermoelectric compound $\text{Mo}_3\text{Sb}_{7-x}\text{Te}_x$

Dipanshu Bansal,^{1,*} Chen W. Li,¹ Ayman H. Said,² Douglas L. Abernathy,³ Jiaqiang Yan,^{1,4} and Olivier Delaire^{1,†}

Te-alloying in $\text{Mo}_3\text{Sb}_{7-x}\text{Te}_x$ shows stiffening of phonon dispersions and DOS.

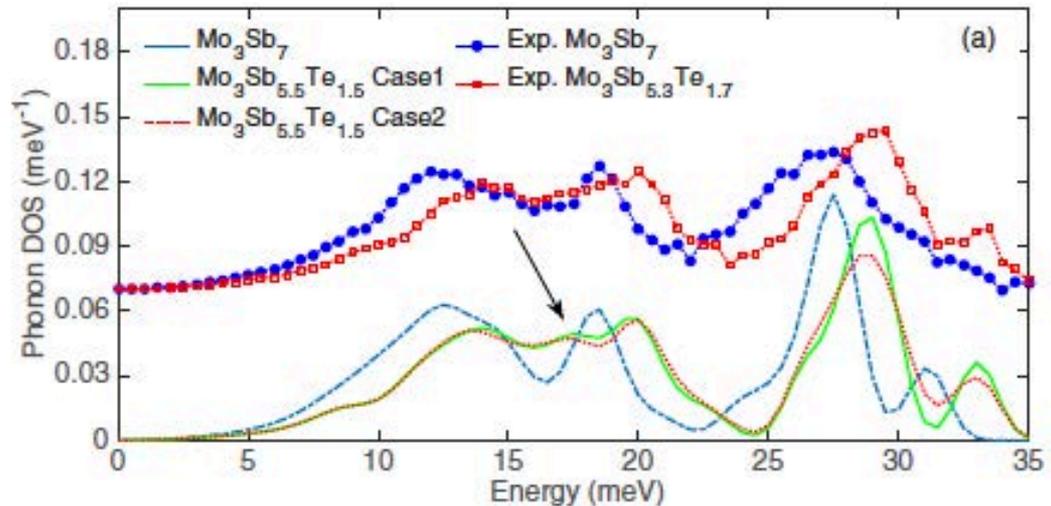
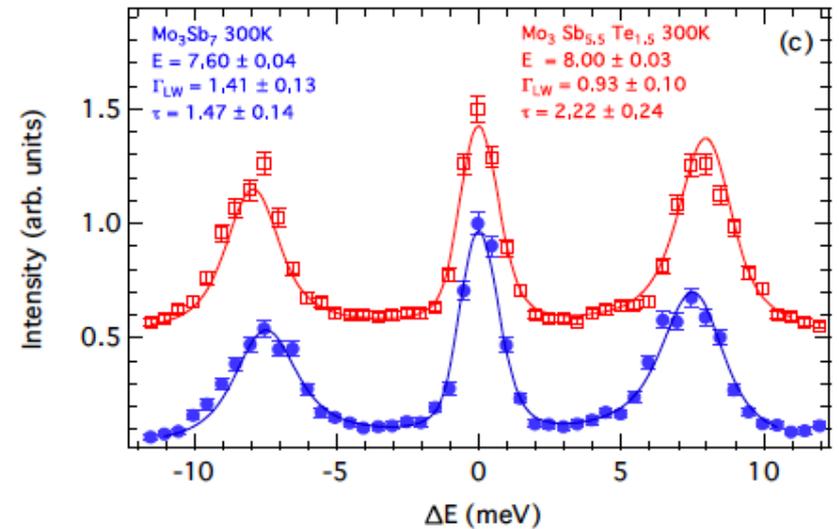
Studied with combination of IXS on small crystals (for dispersions) and INS on powders (for DOS).

Te-alloying fills valence band and decreases electronic density, suppressing electron-phonon coupling.

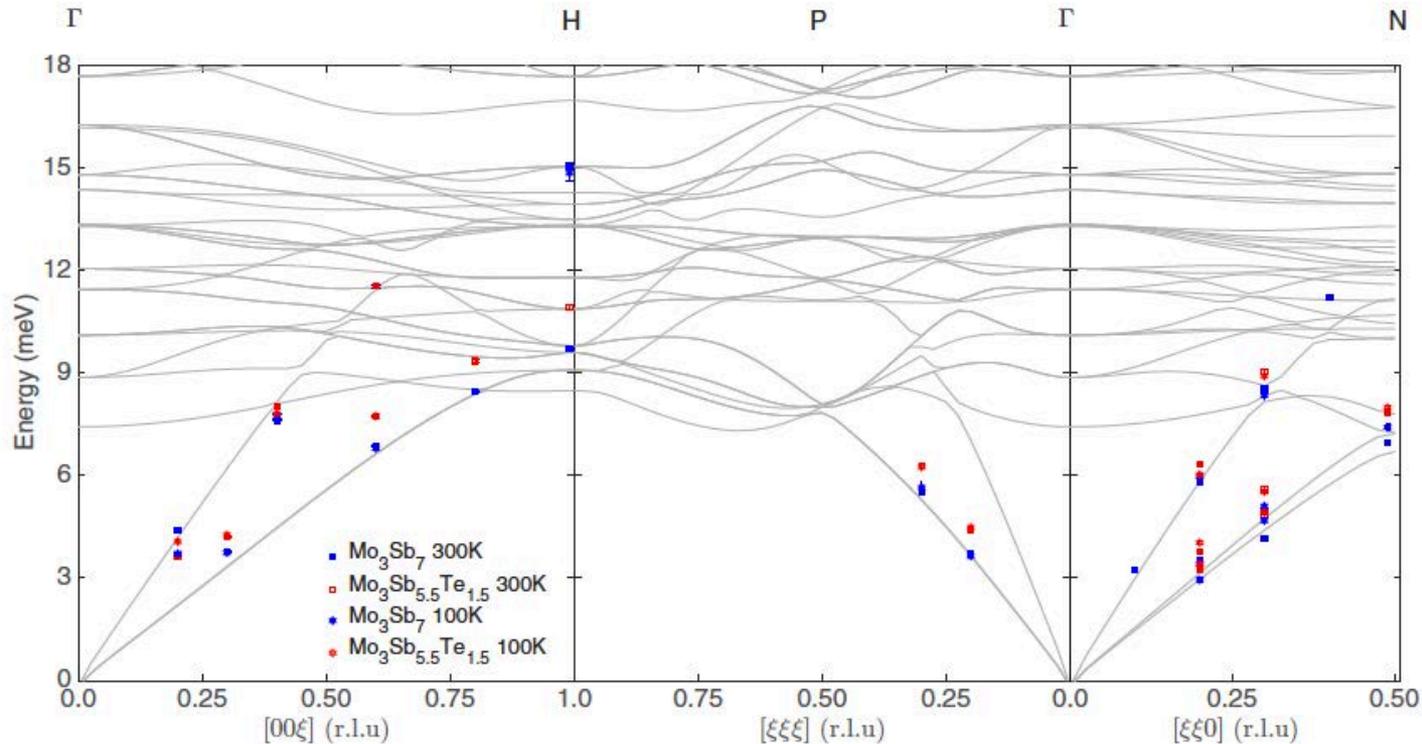


2mm crystal in reflection geometry

Measured on HERI



Comparison with DFT dispersions



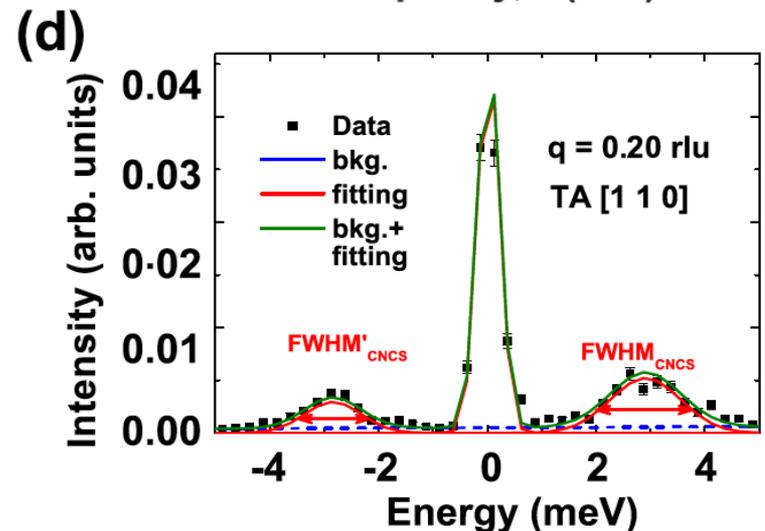
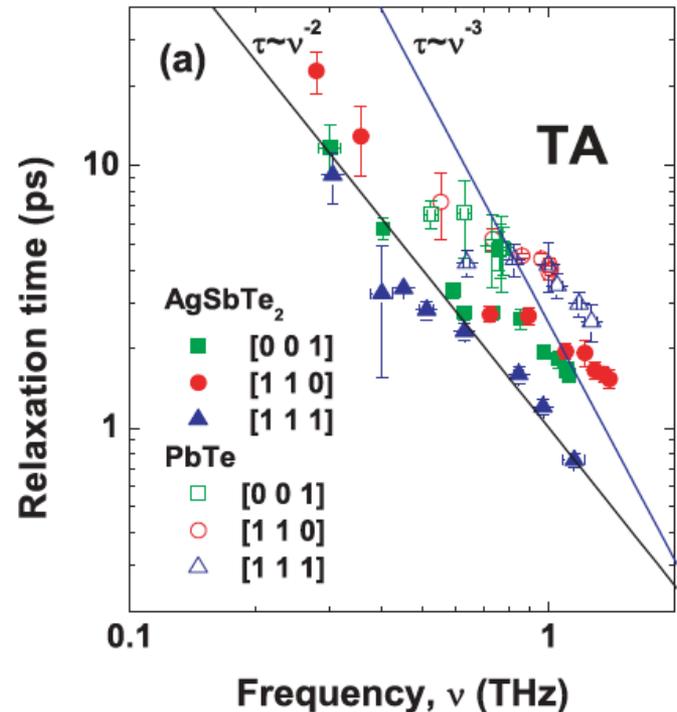
First-principles (DFT) simulations achieve good agreement with IXS

But need experimental data on linewidths to push theory:

combined anharmonicity, electron-phonon, disorder, spin-phonon scattering

Challenges ahead:

- Necessary to analyze many mode linewidths to validate theory: need automatization of resolution corrections
- Need better E-resolution $\sim 10\text{-}50\mu\text{eV}$ for longer phonon lifetimes, in materials with larger thermal conductivities $>10\text{ W}/(\text{m}\cdot\text{K})$
- Monte-Carlo modeling of Q-E resolution and multiple scattering effects with scattering kernels including phonon relaxation (with Jiao Lin).



Conclusions

- Combined IXS / INS experiments and DFT simulations provide powerful insights.
- Opportunities to investigate phonon scattering mechanisms and push theoretical modeling.
- Strong anharmonicity near lattice instabilities in chalcogenide thermoelectrics (displacive transitions with soft-modes), electron-phonon coupling, spin-phonon coupling.
- **High-resolution and high-intensity IXS are *timely and needed*.**

Thank You